

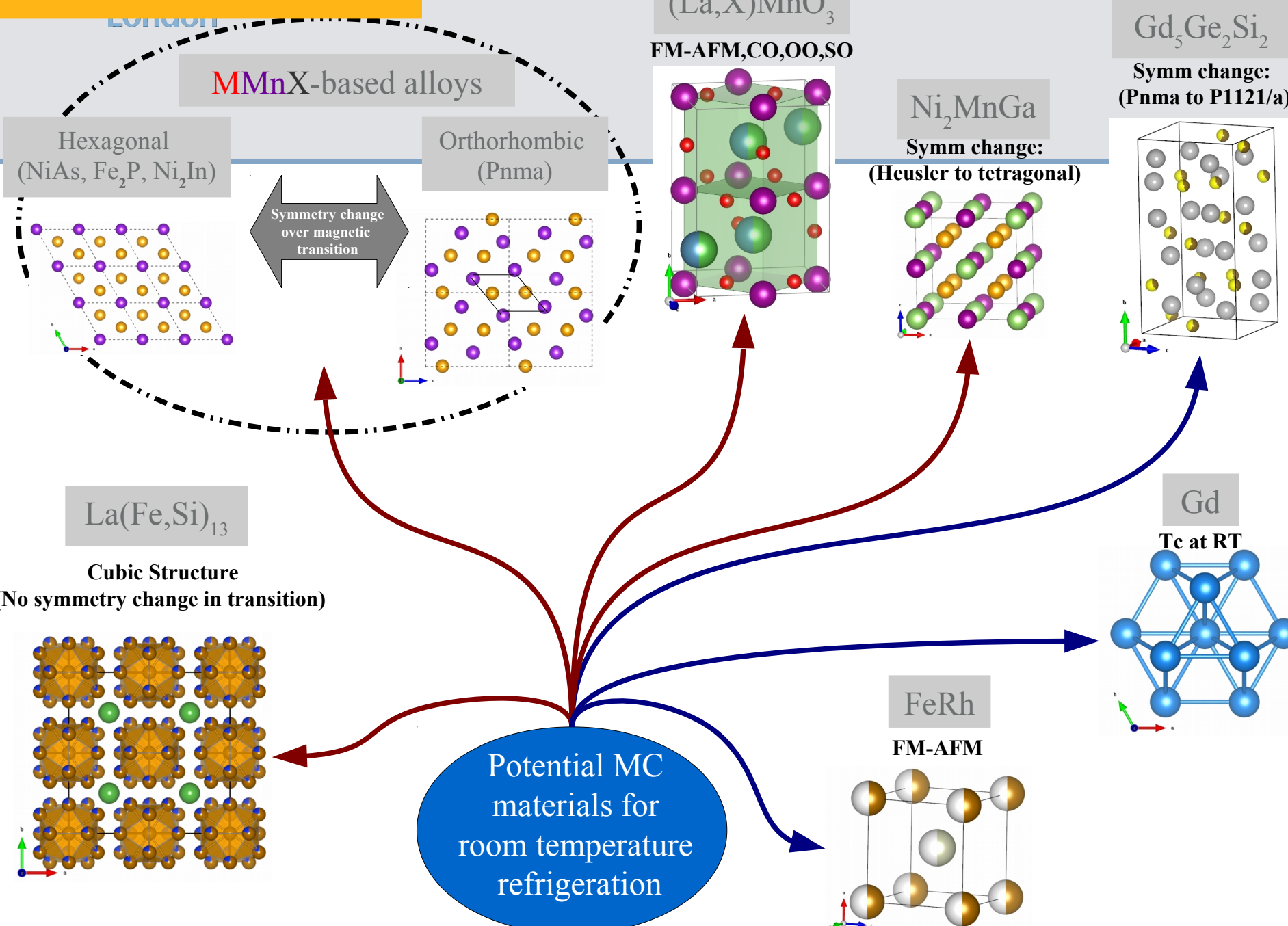
Magnetocaloric Materials Design By Density Functional Theory

Zsolt Gercsi

*Dept. of Physics, Blackett Laboratory
Imperial College London
(Trinity College Dublin)*

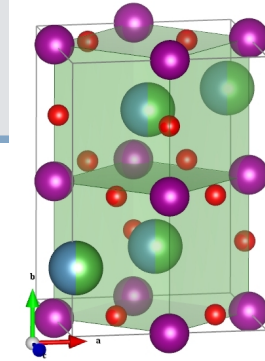
Delft Days on Magnetocalorics (DDMC) 2013

Introduction



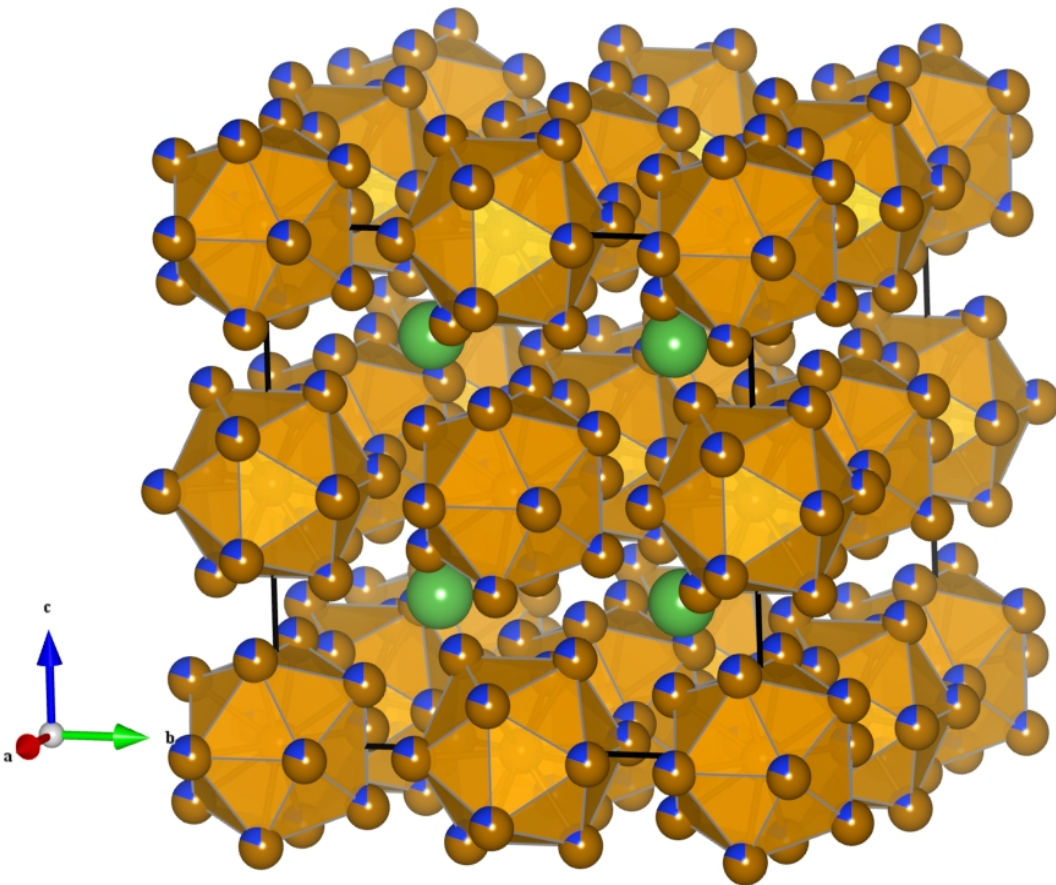


FM-AFM,CO,OO,SO

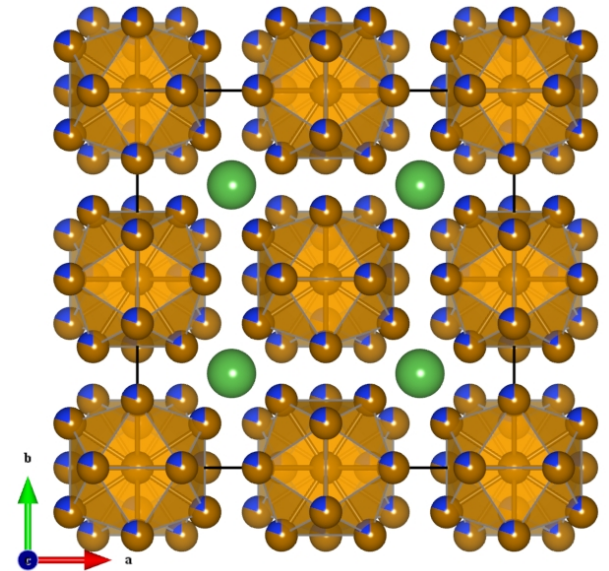


Poster session:
A Hybrid-exchange Density Functional Study
of $\text{La}_{1-x}\text{Ca}_x\text{MnO}_3$
by R. Korotana

Stable Cubic NaZn_{13} type of structure around 10 % Si
(Kripyakewich et al. 1968)

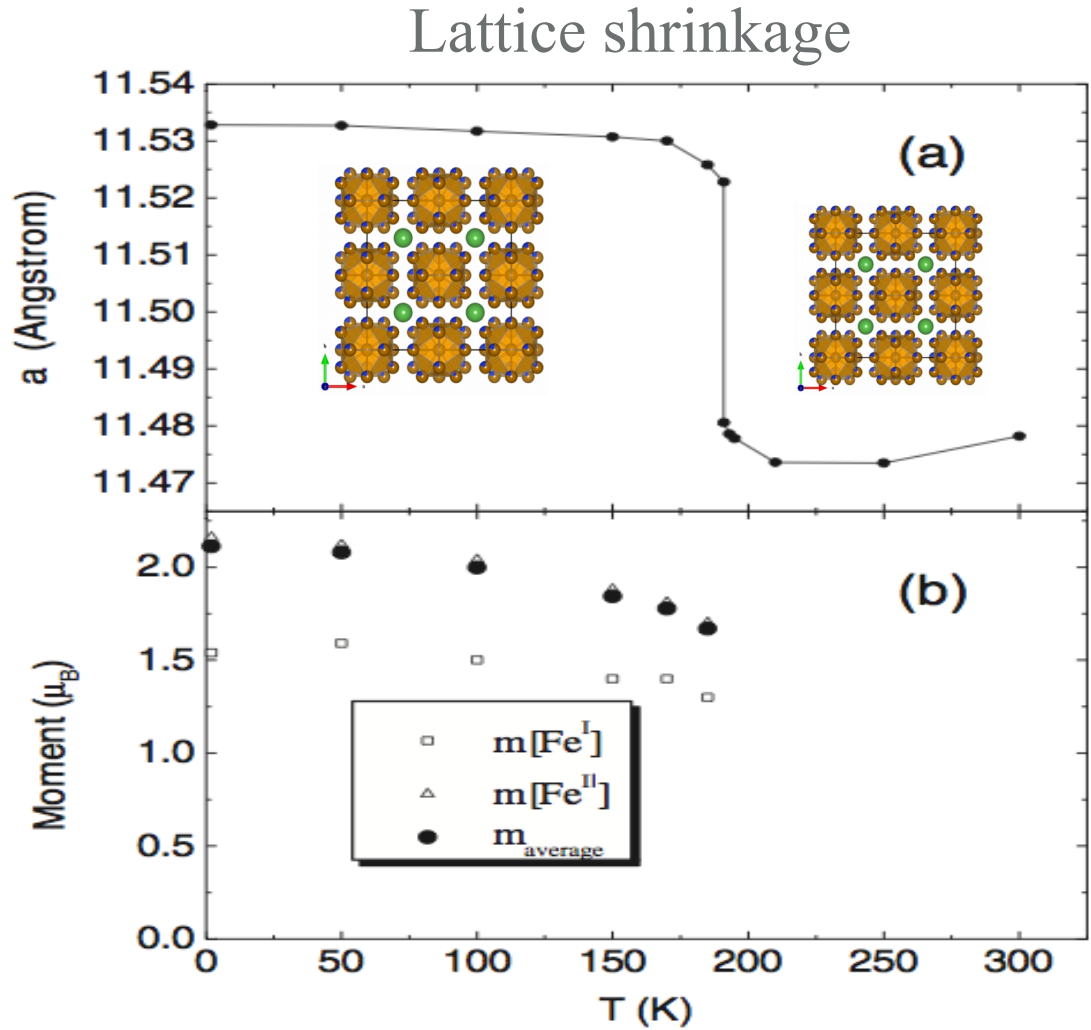


La – 8a sites
Fe (I) – 8b sites
Fe/Si (II) – 96i sites



Fe/Si clusters around La

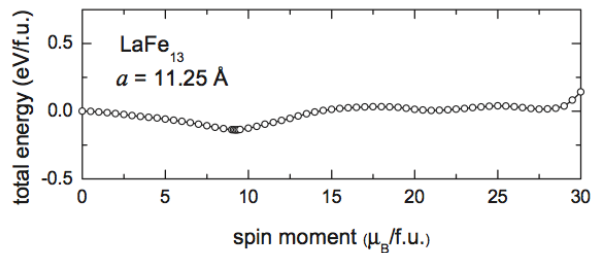
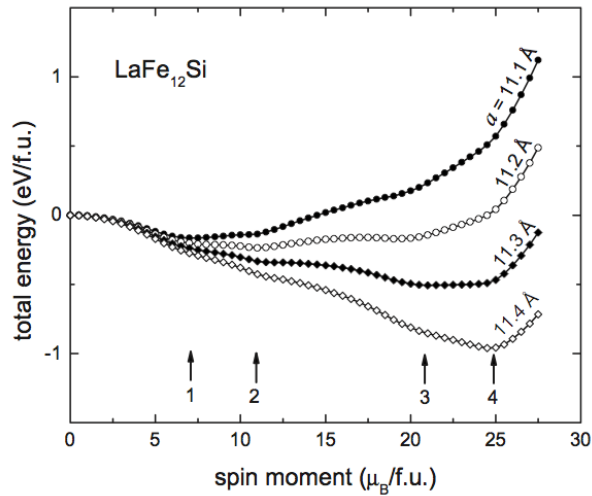
Magnetoelastic transition (no symmetry change, low volume change)



La(Fe,Si)₁₃ - IEM transition

Theory - FPLO-FSM

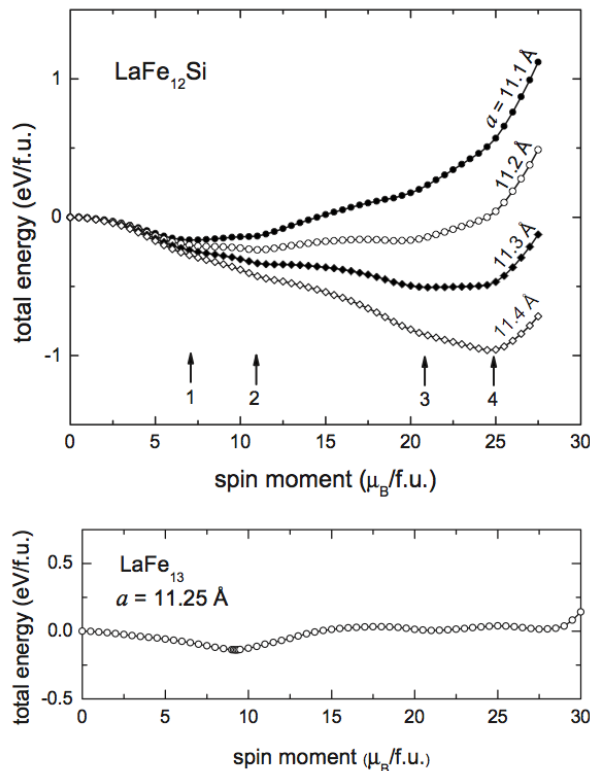
Very shallow energy plateau



La(Fe,Si)₁₃ - IEM transition

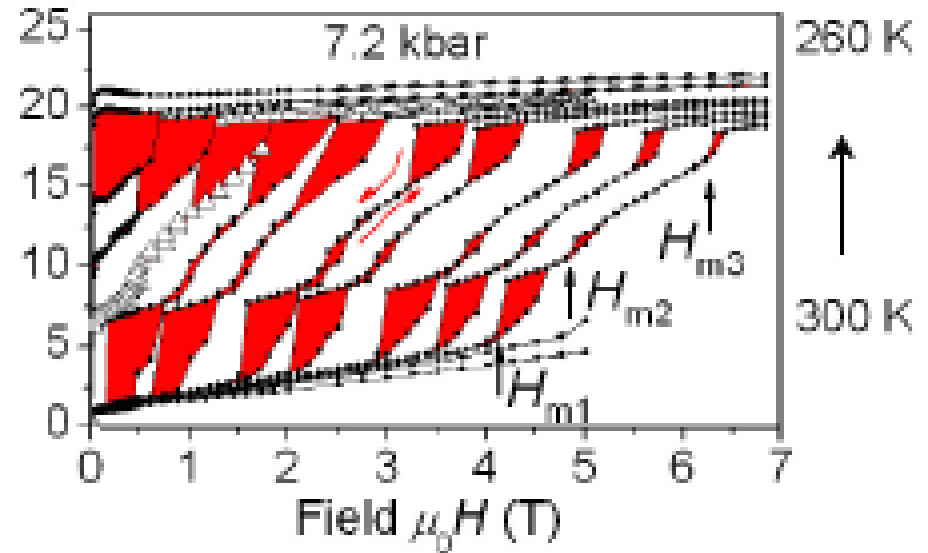
Theory - FPLO-FSM

Very shallow energy plateau



Experimental

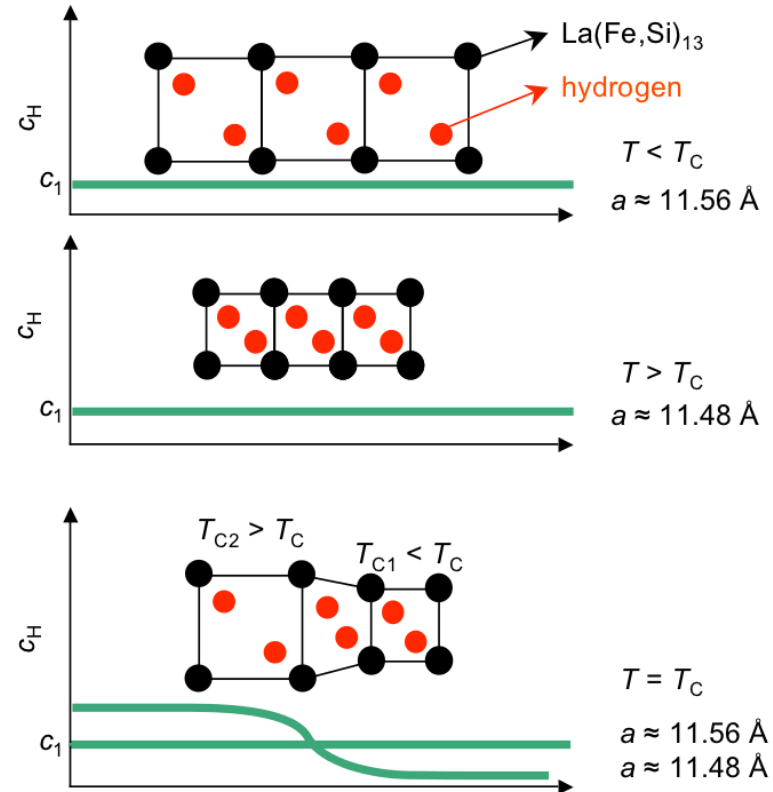
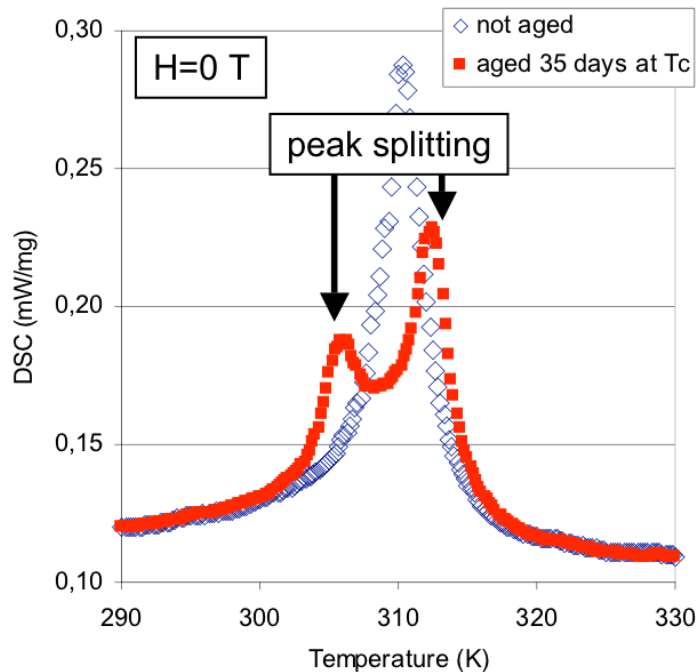
Experimentally observed (??)
under pressure in hydrogenated samples



Hydrogen disproportionation at the Curie temperature

La_{1.04}Fe_{11.44}Si_{1.56}H_{1.35}:

- Produced by induction melting.
- Hydrogenated at 514 K.
- Stored in air at T_C for 35 days.



A. Barcza et al. IEEE Trans Magn, 47 10 (2011)

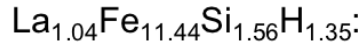
Also see in: M. Krautz et al. J. Appl. Phys. 112, 083918 (2012)

C. B. Zimm et al. J. Appl. Phys. 113, 17A908 (2013)

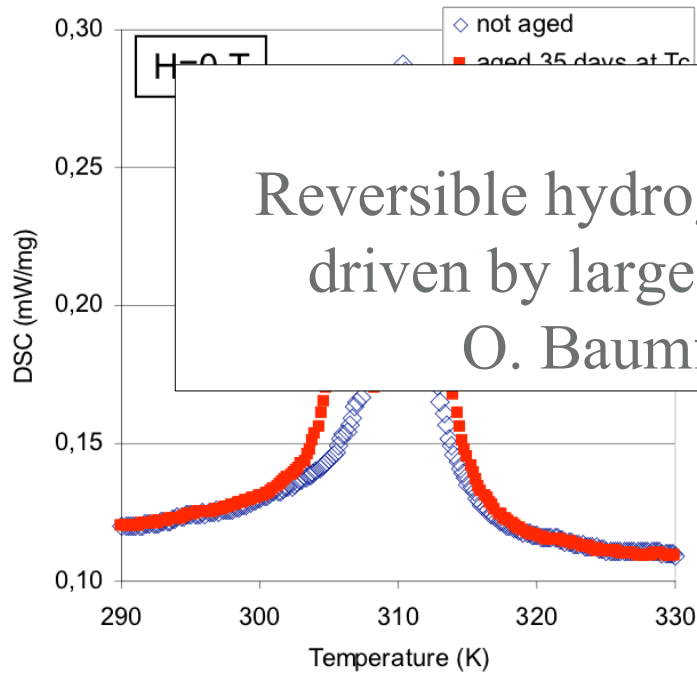
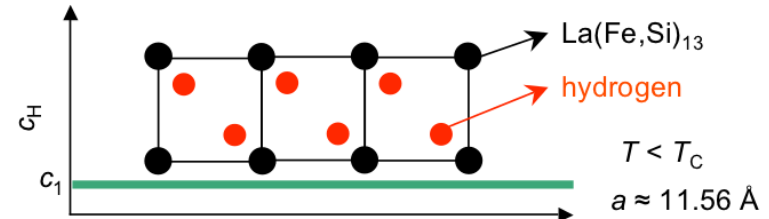
La(Fe,Si)₁₃ - IEM transition

Experimental

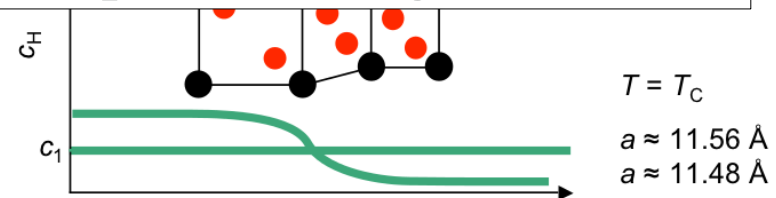
Hydrogen disproportionation at the Curie temperature



- Produced by induction melting.
- Hydrogenated at 514 K.
- Stored in air at T_C for 35 days.



Poster session:
Reversible hydrogen diffusion in LaFe_{13-x}Si_xH_y
driven by large spontaneous magnetostriction
O. Baumfeld – Imperial College



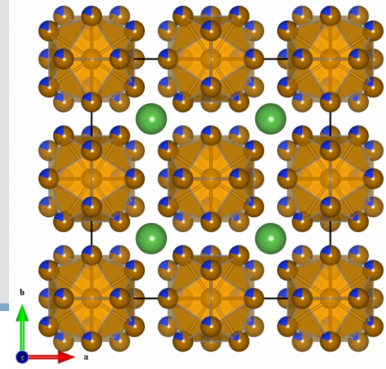
A. Barcza et al. IEEE Trans Magn, 47 10 (2011)

Also see in: M. Krautz et al. J. Appl. Phys. 112, 083918 (2012)

C. B. Zimm et al. J. Appl. Phys. 113, 17A908 (2013)

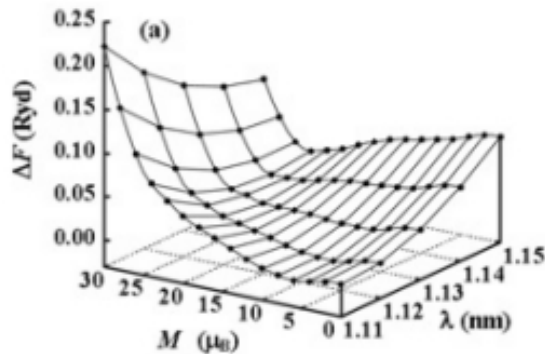
La(Fe,Si)₁₃ - DFT

La – 8a sites
Fe (I) – 8b sites
Fe/Si (II) – 96i sites

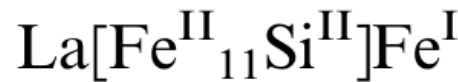
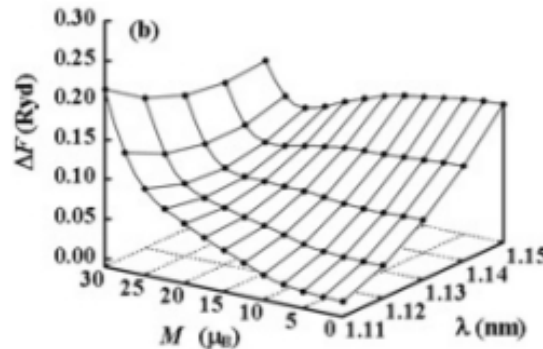


The calculated multi-minima structure depends strongly on the theory used as well as on the Si occupation considered.

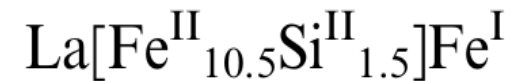
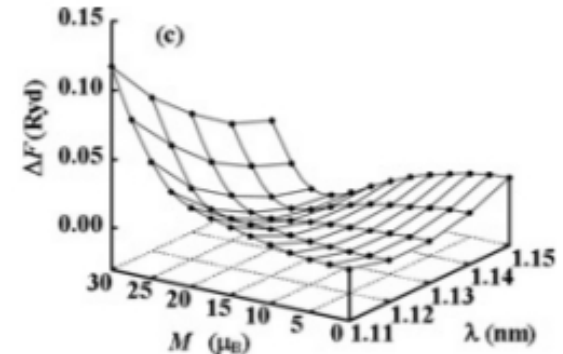
A. Fujita, H. Yakoi / Scripta Materialia 67 (2012) 578–583



The multiplicity of transition fades and concave–convex variation appears (M=5μ_B state is stable).

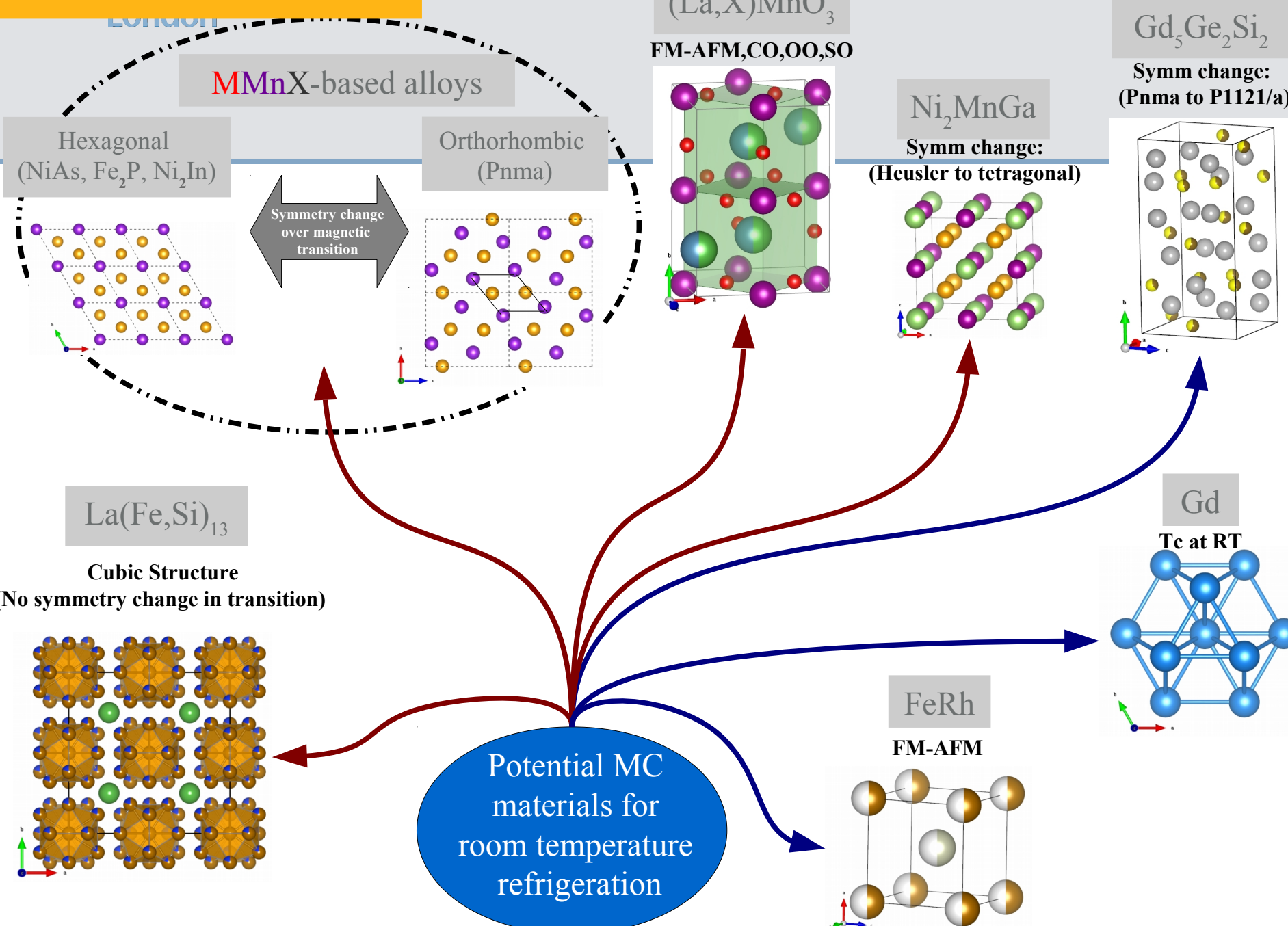


Si occupation of the 96i site show gives stable 0μ_B non-magnetic state.



CPA simulations gives the FM state most stable.

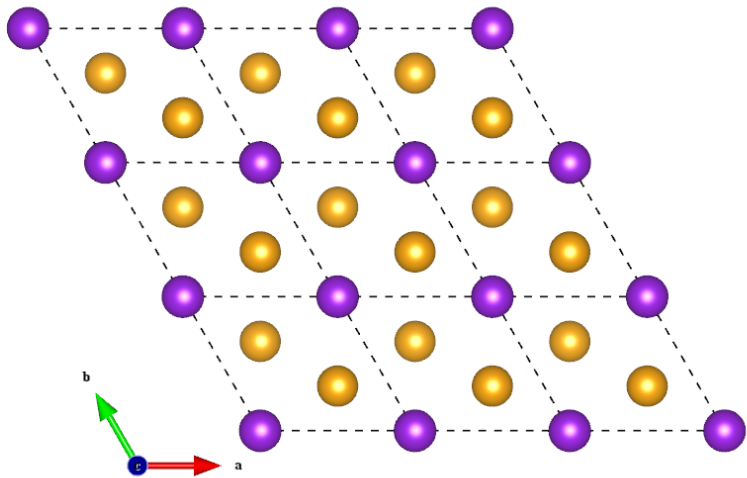
Introduction



MMnX-based metallic alloys

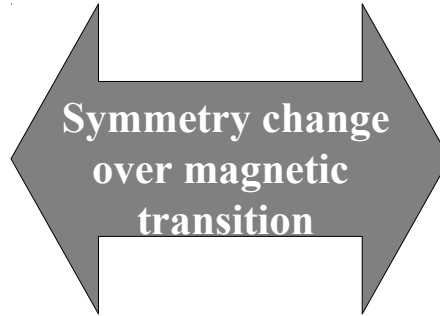
M – void or d-block element

X – p-block element



Hexagonal
(Fe₂P, NiAs, Ni₂In)

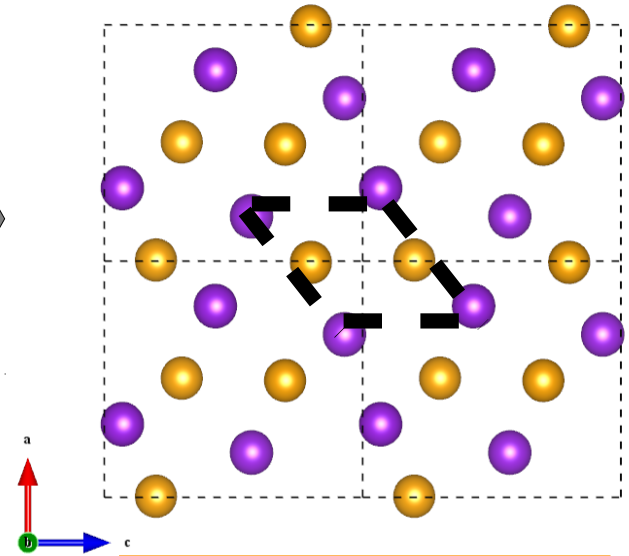
No symmetry change



MnAs



...



Orthorhombic
(Pnma)

No symmetry change



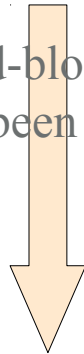
Magnetostructural transitions – MnAs

From NiAs to Pnma structure \longrightarrow

• MnAs

Wada et. al. Physica B: Cond. Mat. 328, 1, (2003)

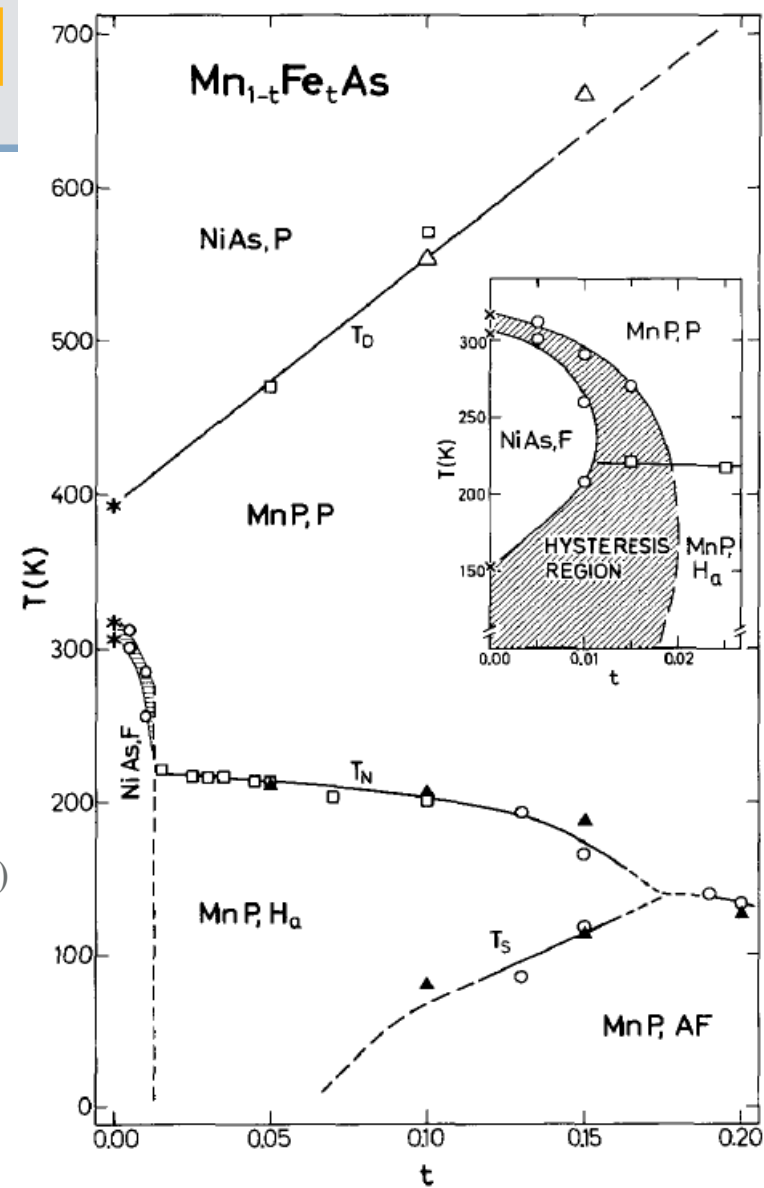
Both p- and d-block type dopants
have been explored



• $Mn_{1-x}Fe_xAs$, $MnAs_{1-x}Sb_x$, etc.

See for instance in De Campos et al Nature Materials 5, 802 – 804 (2006)

Relevant DFT study?



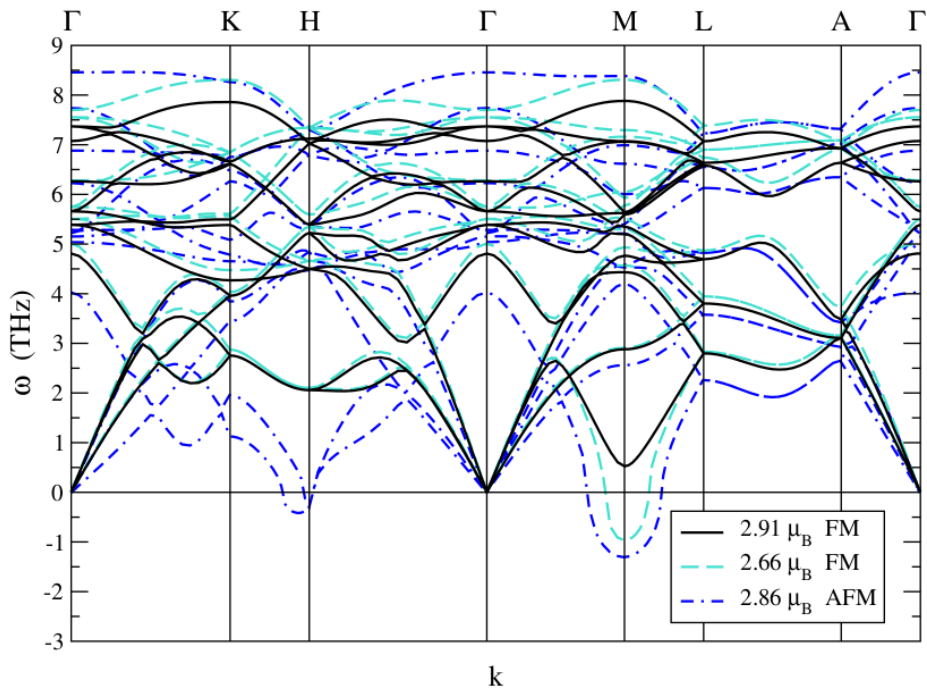
Fjellvag et. al. JMMM 46, 29, (1984)

Magnetostructural transitions – MnAs

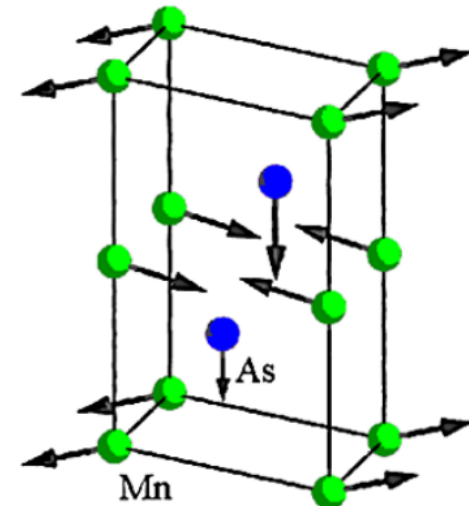
DFT

Strong (giant) coupling
between **lattice and magnetic** interactions

Phonon dispersion of MnAs (hexagonal)



Mn atoms move in the hexagonal a-b plane in one direction, while As atoms move along the c-direction



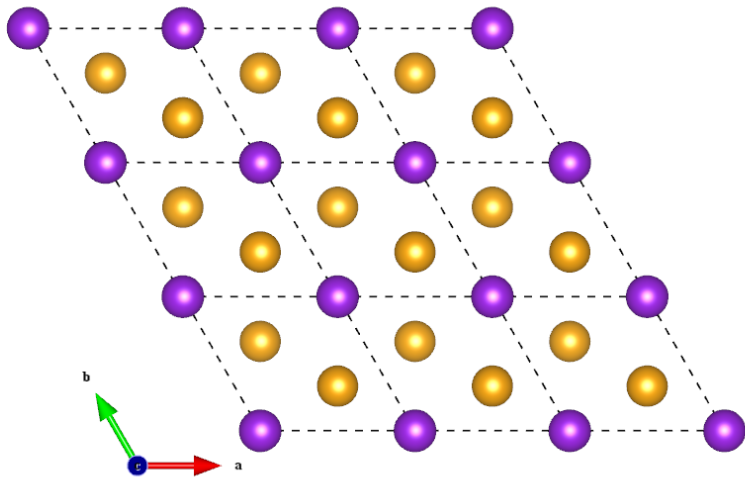
consistent with experimental findings
(neutron diffraction)

For more, see in
J. Łazewski et al., PRL 104, 147205 (2010)

MMnX-based metallic alloys

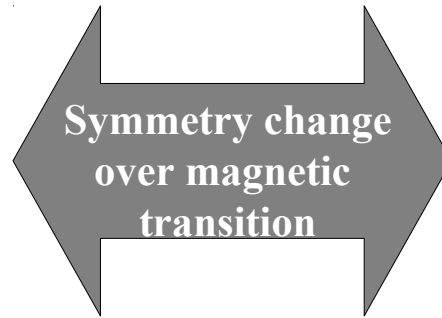
M – void or d-block element

X – p-block element

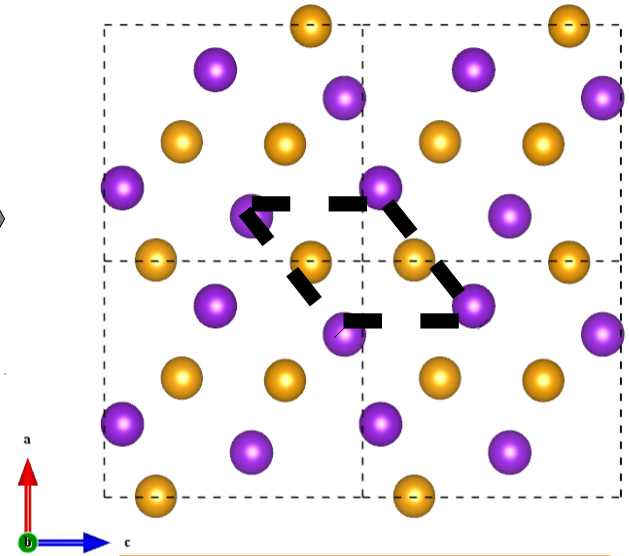


Hexagonal
(Fe₂P, NiAs, Ni₂In)

No symmetry change



...



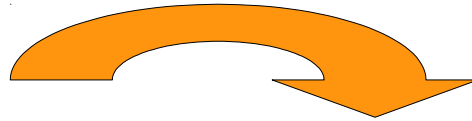
Orthorhombic
(Pnma)

No symmetry change

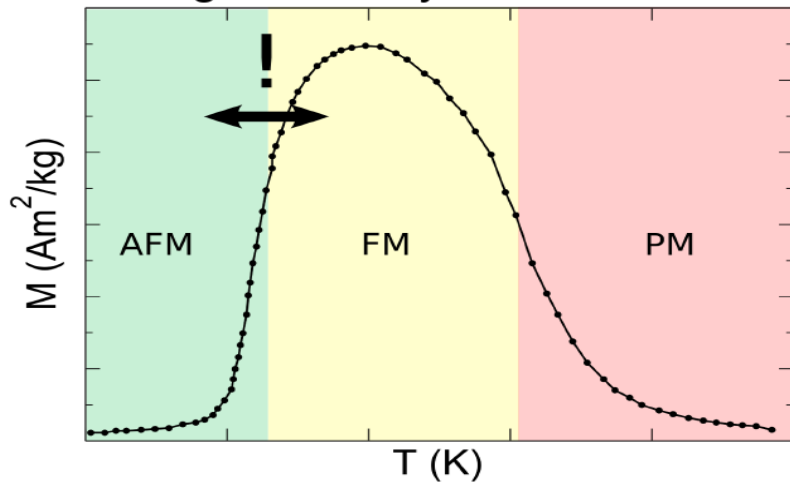


Orthorhombic (Pnma) metamagnet:

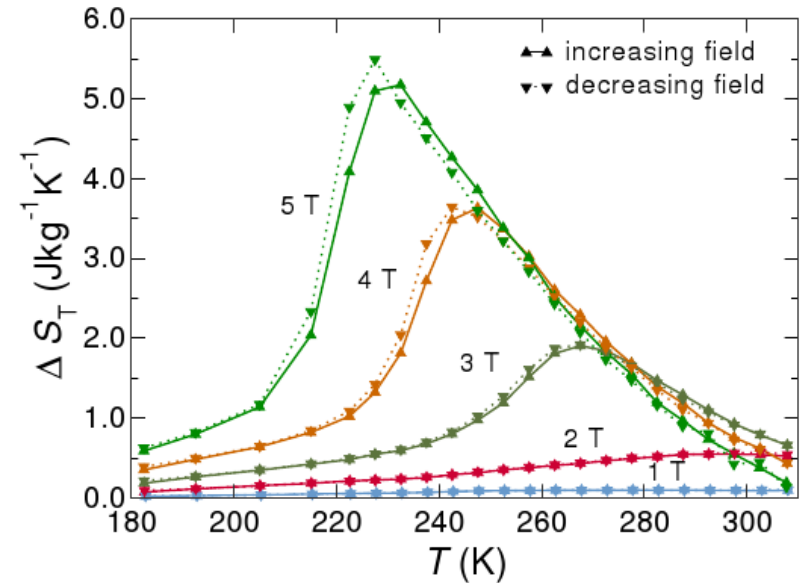
CoMnSi



Bulk magnetometry:

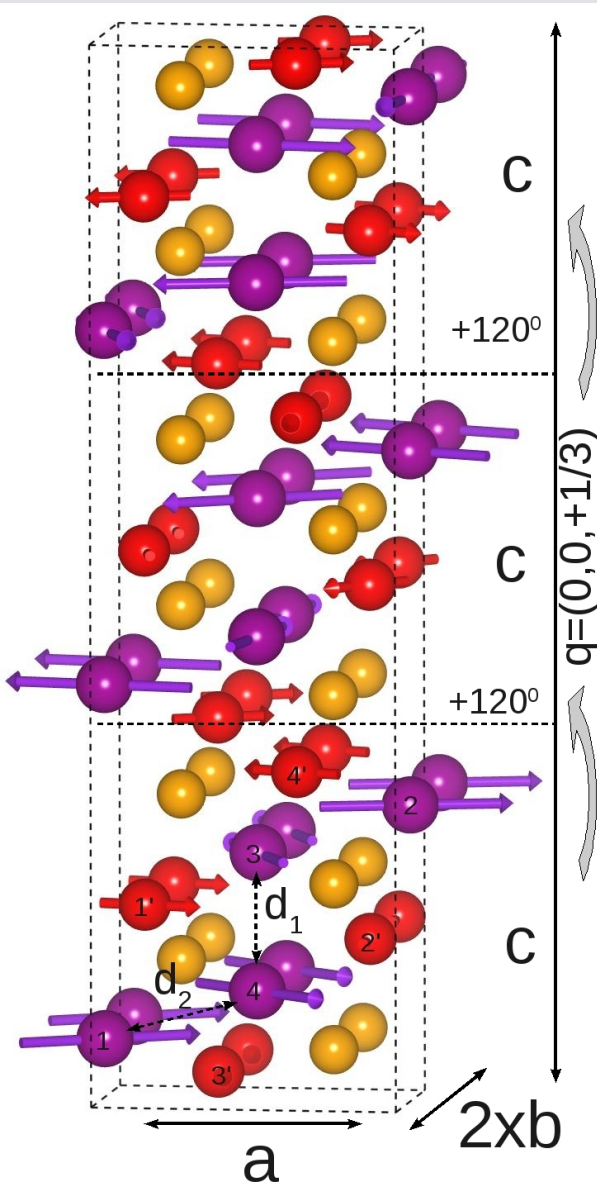


after H. Binczycka, A. Szytula, Phys. Stat. Sol. A, **35** K69-K72 (1976)

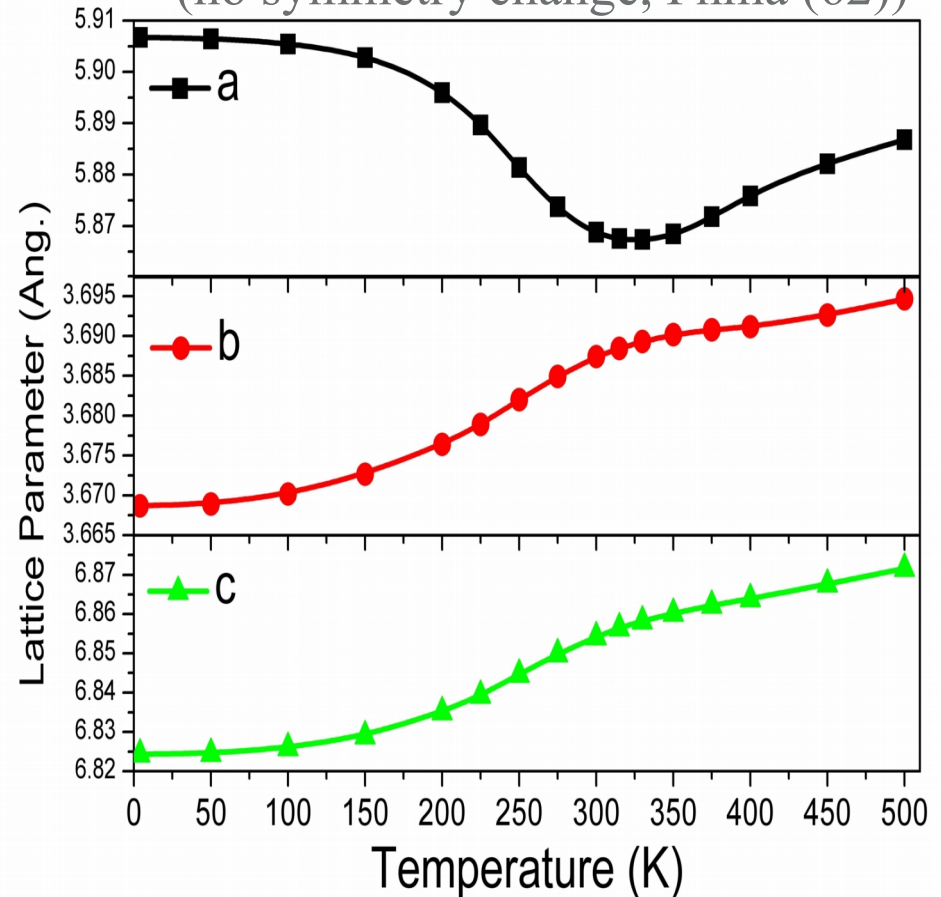


K.G. Sandeman et al.
Phys. Rev. B 74, 224436 (2006)

$$\mu_{\text{Mn}} \approx 2.6\mu_{\text{B}}, \mu_{\text{Co}} \approx 0.3\mu_{\text{B}}$$



Negative a -lattice parameter expansion
(no symmetry change, Pnma (62))

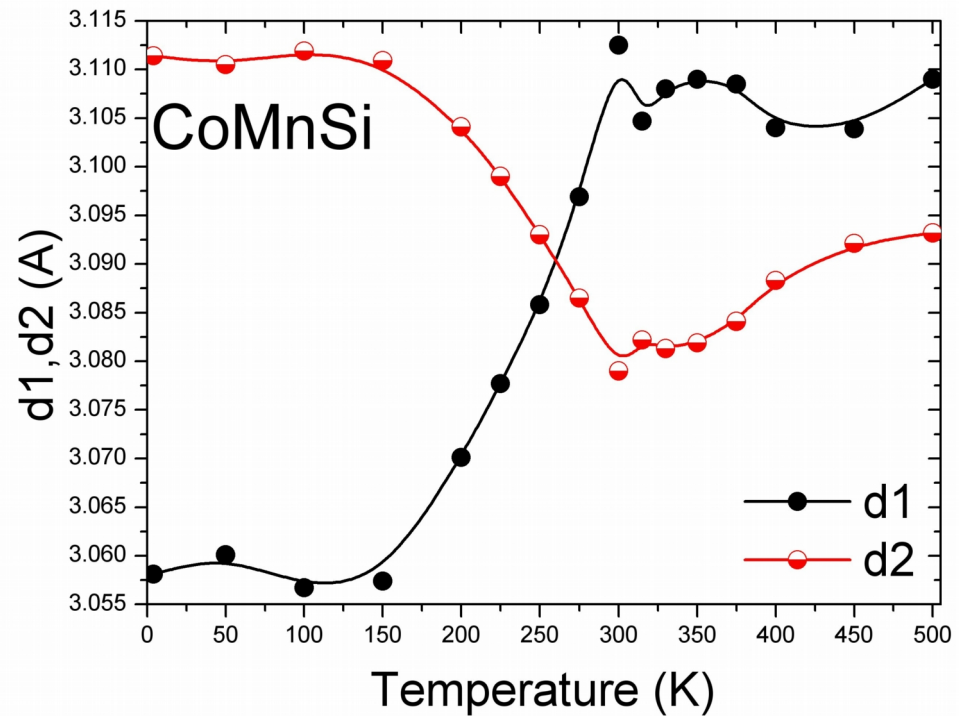
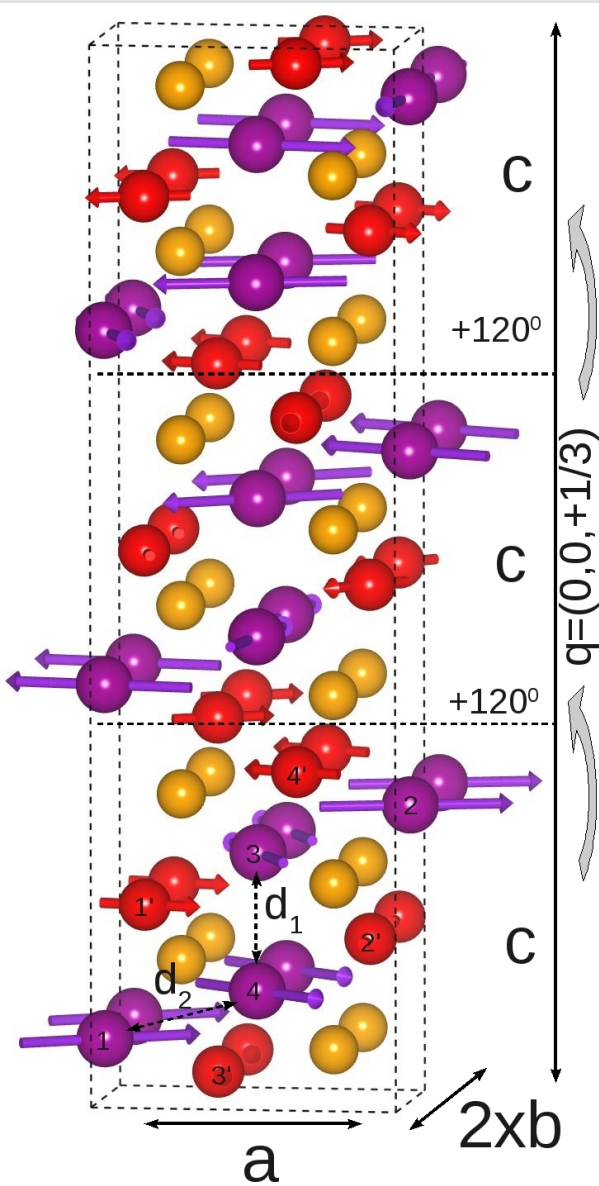


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CoMnSi

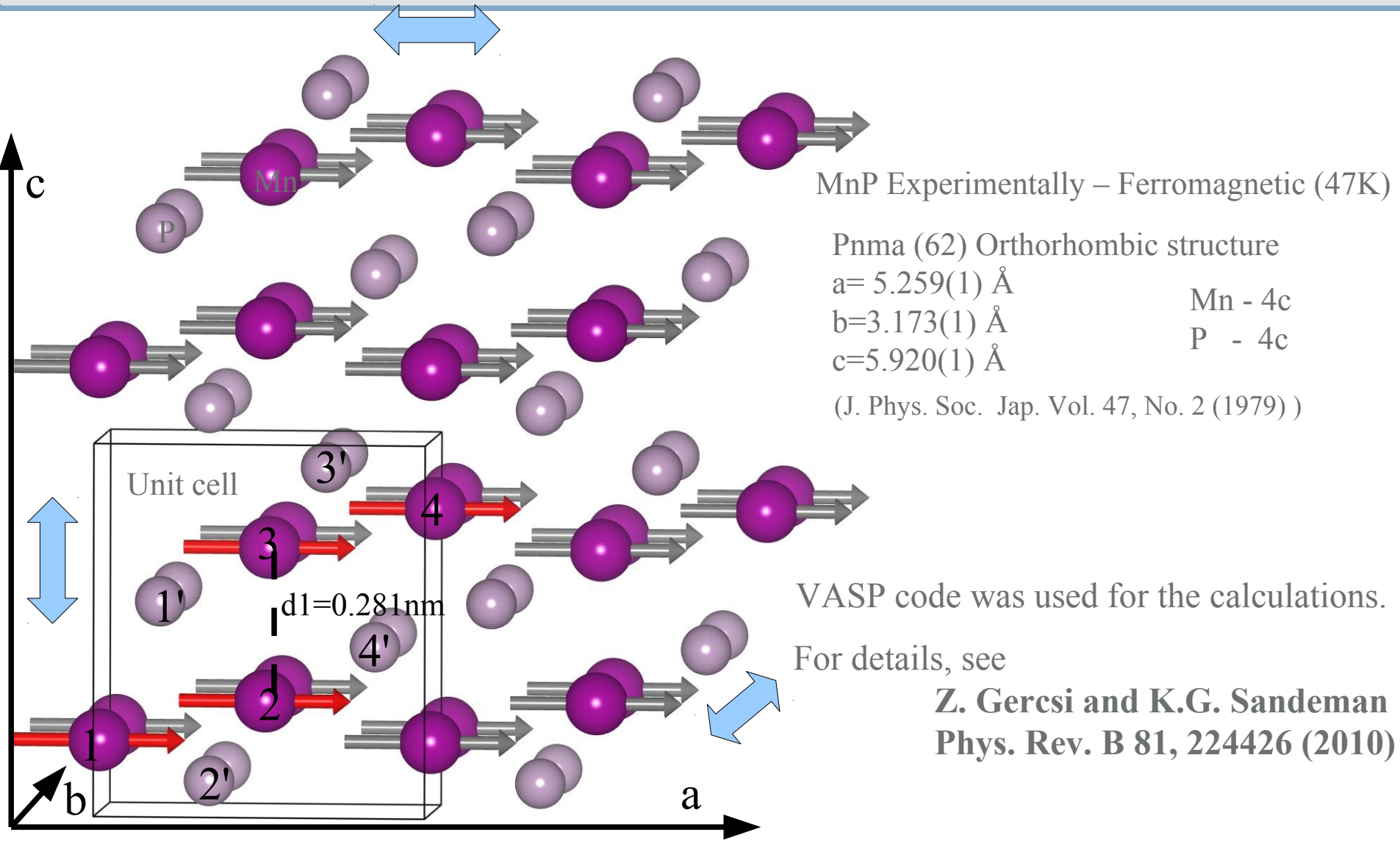
High Resolution Powder Diffraction (ISIS, Didcot, UK)

We found giant changes in Mn-Mn distances (d_1, d_2) link to the metamagnetic transition.

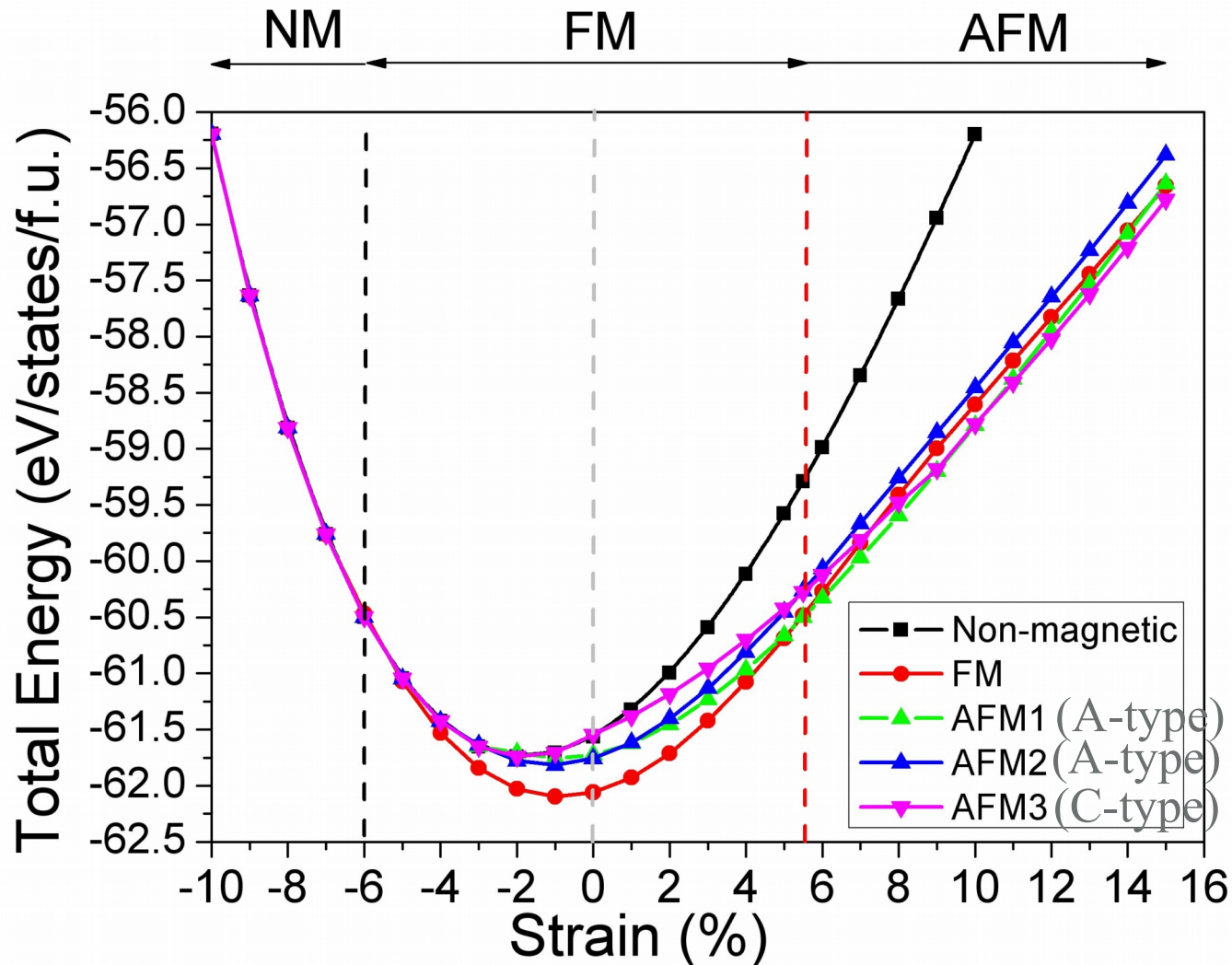


A. Barcza, Z. Gercsi, K.S. Knight and K.G. Sandeman
Phys. Rev. Lett. 104, 247202 (2010)

DFT theory to map the magnetic phase stability
vs. lattice volume using the prototype (MnP) structure

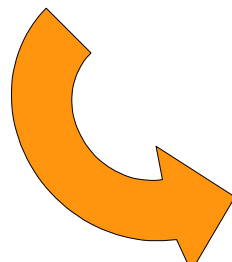


DFT theory to map the magnetic phase stability
vs. lattice volume using the prototype (MnP) structure



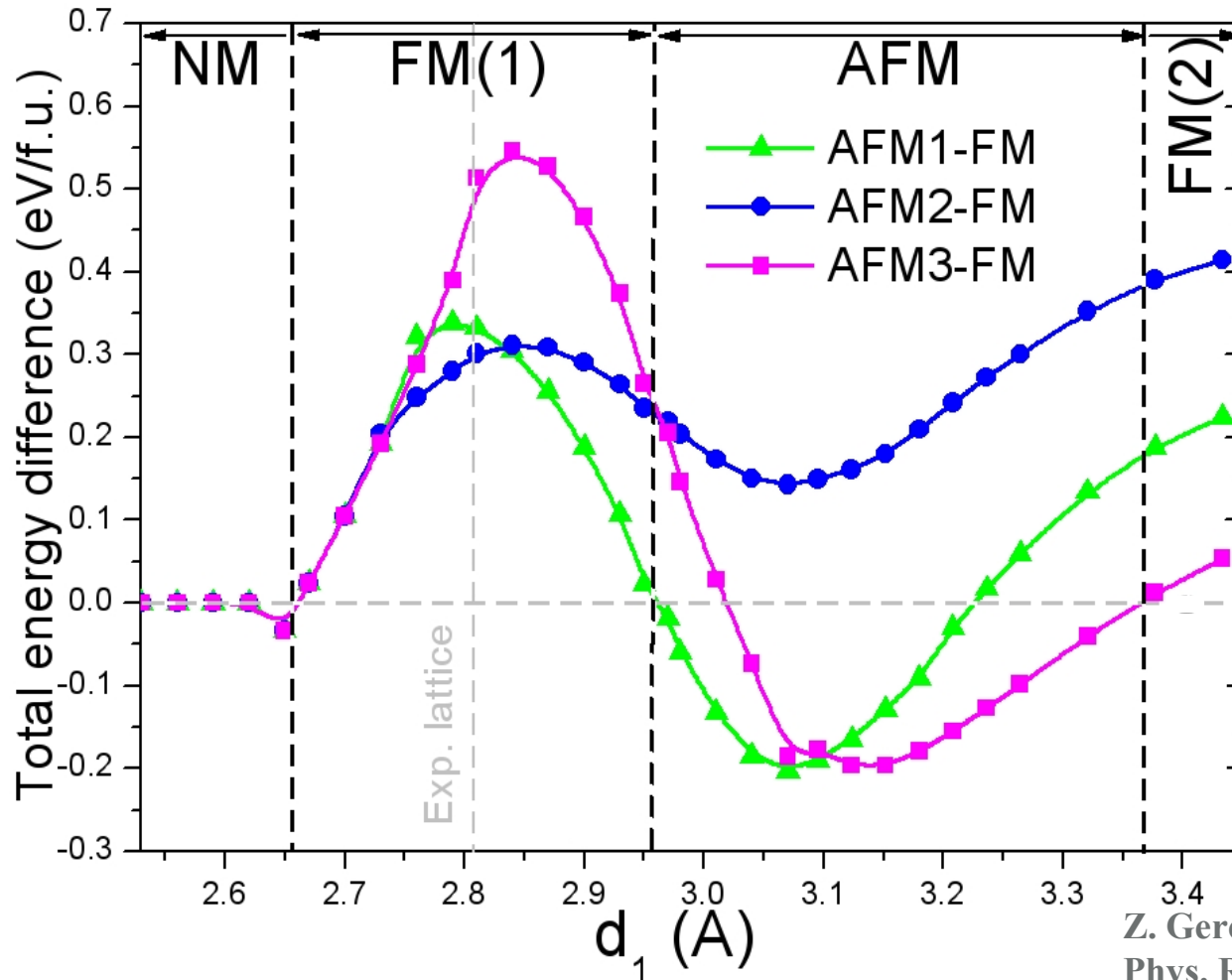
- Total energy curve predicts ferromagnetic ground state in accordance with experimental.

- These energies can also be plotted with respect to the FM phase using a more informative (Mn-Mn interatomic) scale.



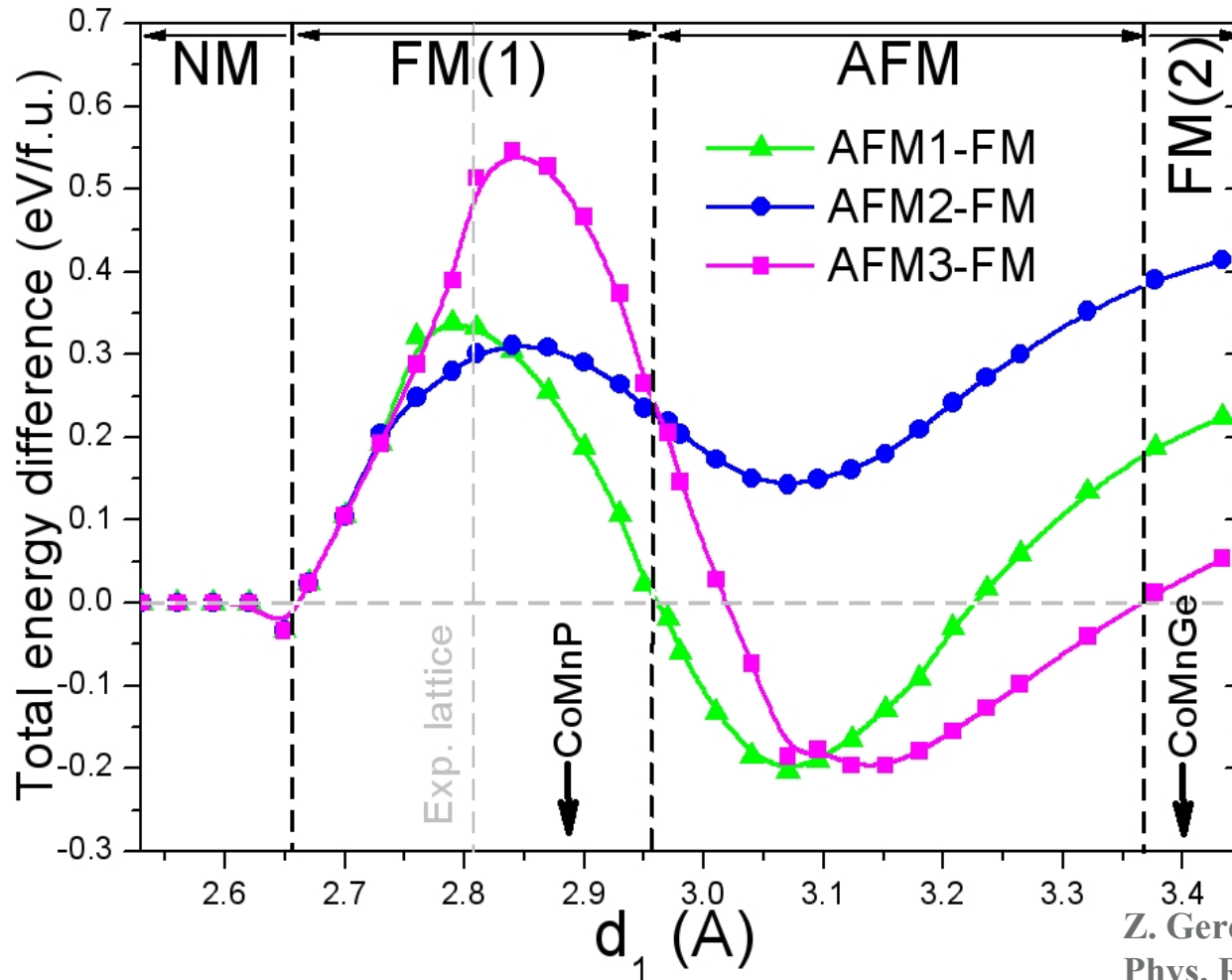
New metamagnet from DFT: CoMn(GeP)

The stability plot predicts stable AFM ground state for $\text{CoMnGe}_{1-x}\text{P}_x$ with $x \sim 0.5$.



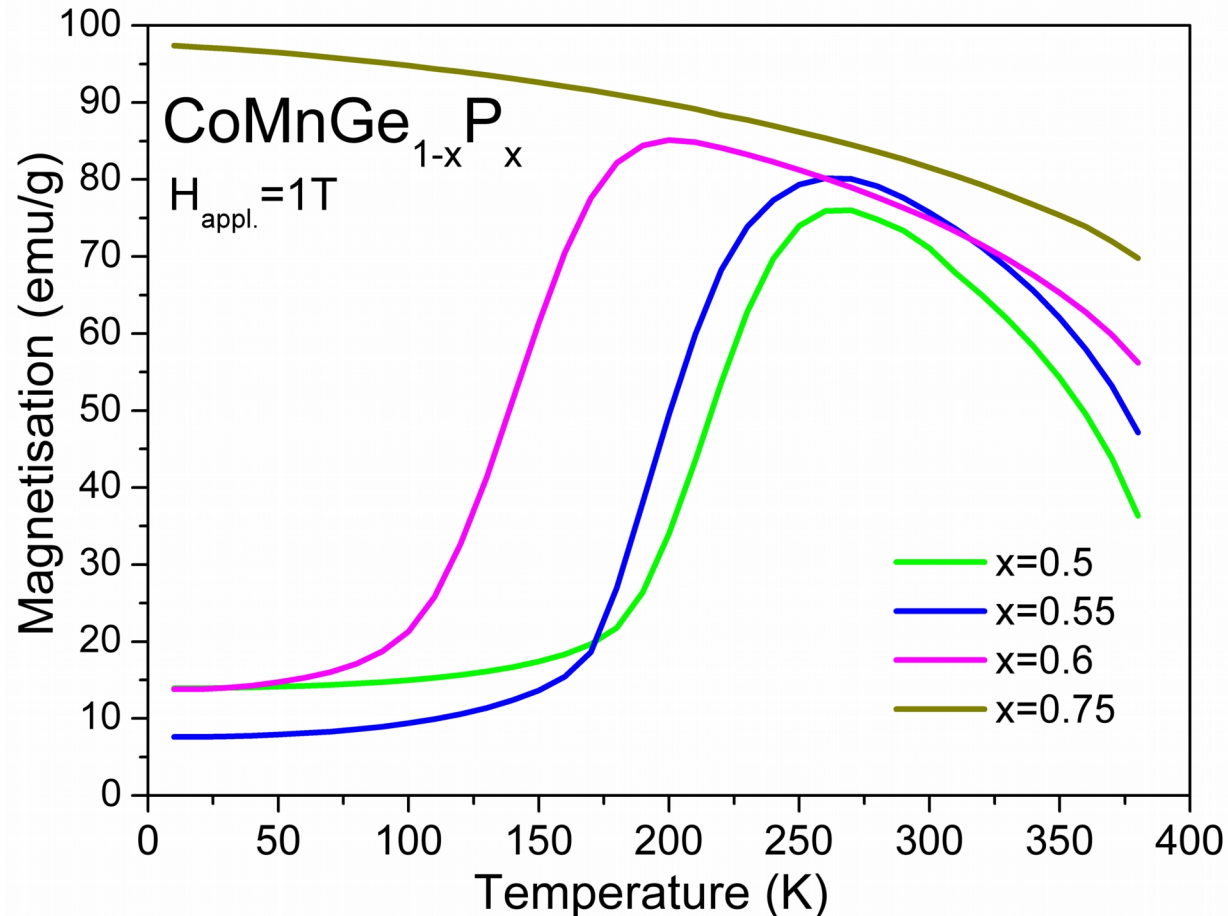
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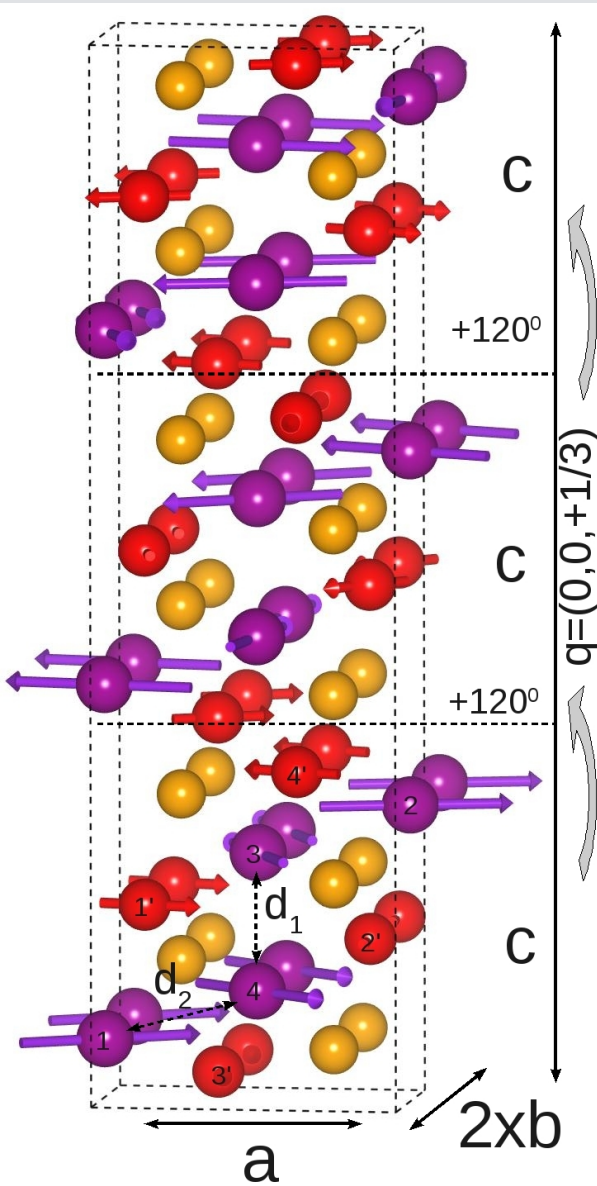


New metamagnet from DFT: CoMn(GeP)

AFM ground state with metamagnetism for $x=0.5, 0.55, 0.6$ as predicted!



$$\mu_{\text{Mn}} \approx 2.6\mu_{\text{B}}, \mu_{\text{Co}} \approx 0.3\mu_{\text{B}}$$



Our recent works on these materials:

A. Barcza, Z. Gercsi, et al. Phys. Rev. Lett. **104**, 247202 (2010)

Z. Gercsi and K.G. Sandeman Phys. Rev. B **81**, 224426 (2010)

Z. Gercsi, K. Hono and K.G. Sandeman Phys. Rev. B **83**, 174403 (2011)

A. Barcza, Z. Gercsi et al. Phys. Rev. B **87**, 064410 (2013)

Q. Recour, V. Ban, Z. Gercsi et al. Phys. Rev. B **88**, 054429 (2013)

J. B. Staunton, M. dos Santos Dias, J. Peace, Z. Gercsi, and K. G. Sandeman
Phys. Rev. B **87**, 060404 (2013)

09:30 – 10:00

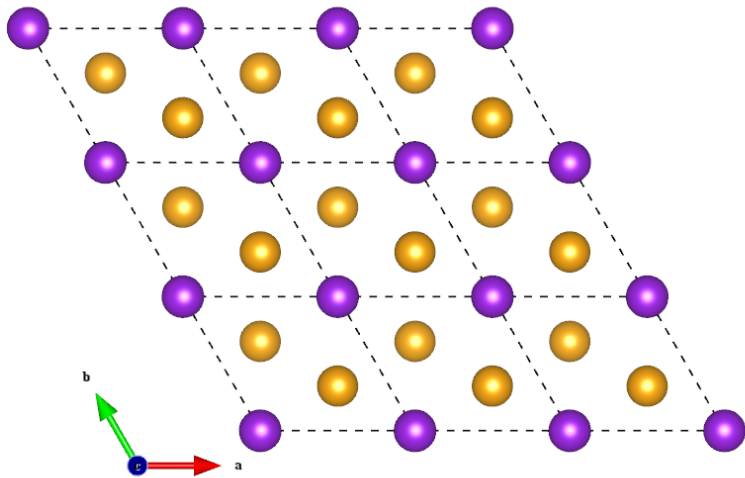
Tuning the metamagnetism of an
antiferromagnetic metal

J. Staunton – Warwick University

MMnX-based metallic alloys

M – void or d-block element

X – p-block element



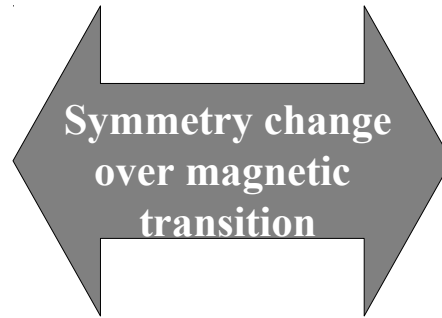
Hexagonal
(Fe₂P, NiAs, Ni₂In)

No symmetry change

Fe₂P

MnFeP_{1-x}As_x

MnFeP_{1-x}Si_x, ...



MnAs

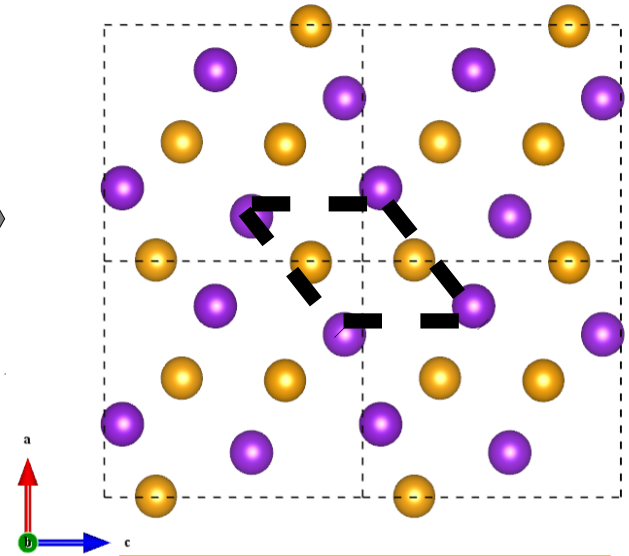
MnAs_{1-x}Sb_x

Mn_{1-x}Fe_xAs

Co_{1-x}MnGe

CoMnGeB_x

...



Orthorhombic
(Pnma)

No symmetry change

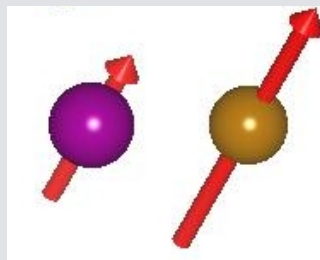
CoMnSi

NiMnGe_{1-x}Si_x

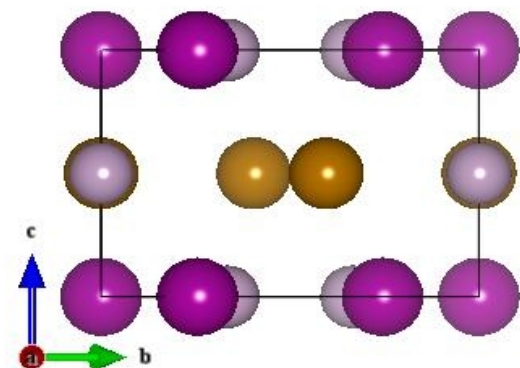
Mn₃Sn₂, ...

Parent \rightarrow Fe_2P

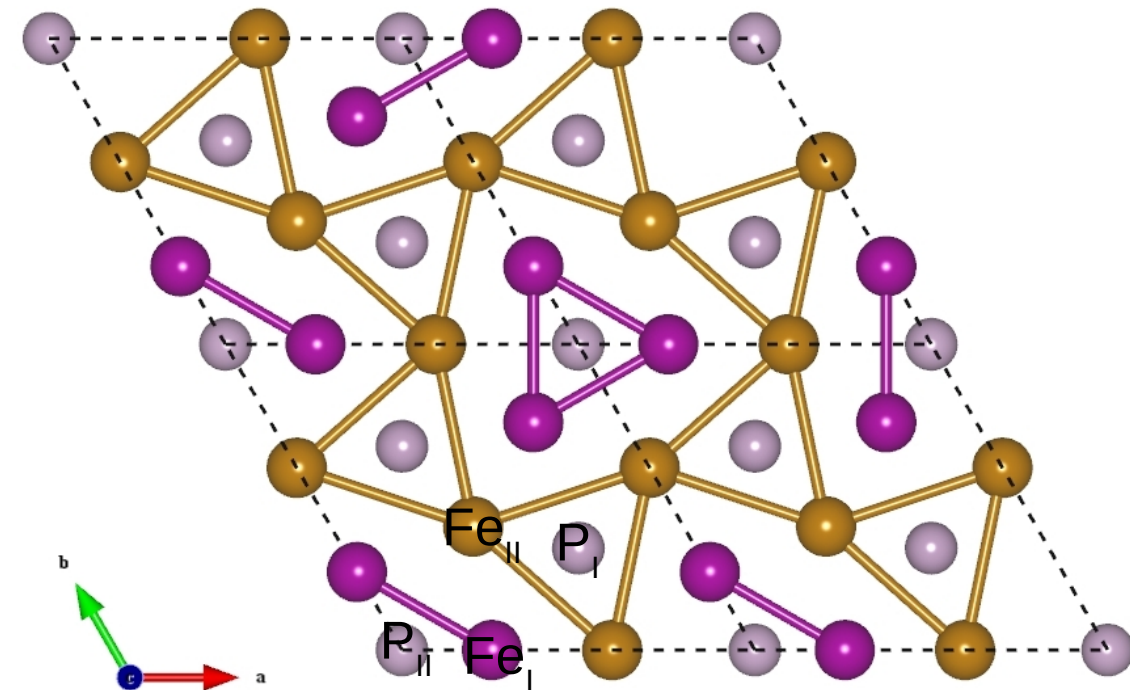
P-62m (189)



Fe_I	3f sites $\sim 0.8\mu_B$
Fe_{II}	3g sites $\sim 2.4\mu_B$
P	1b&2c sites

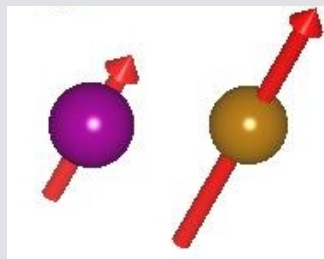


Fe_2P is the prototype of the space group with unusual magnetic properties.

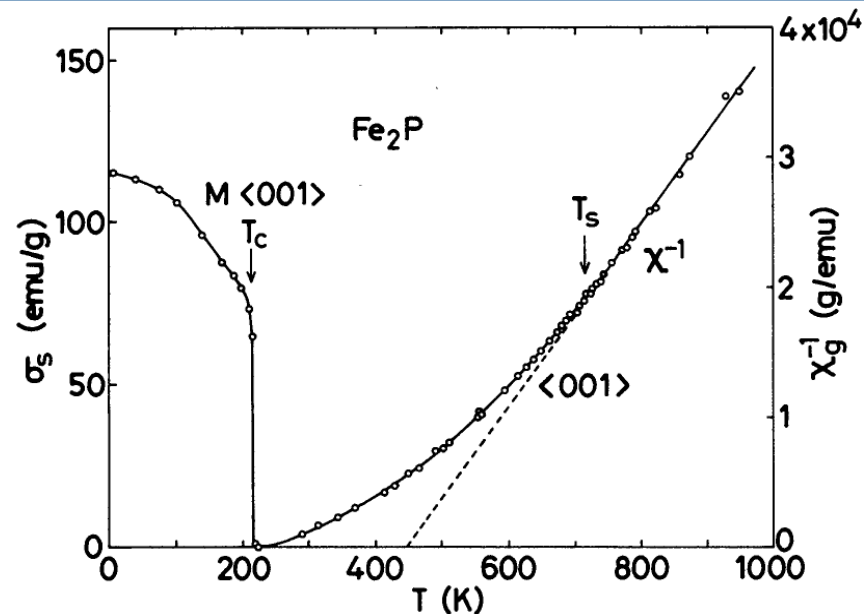
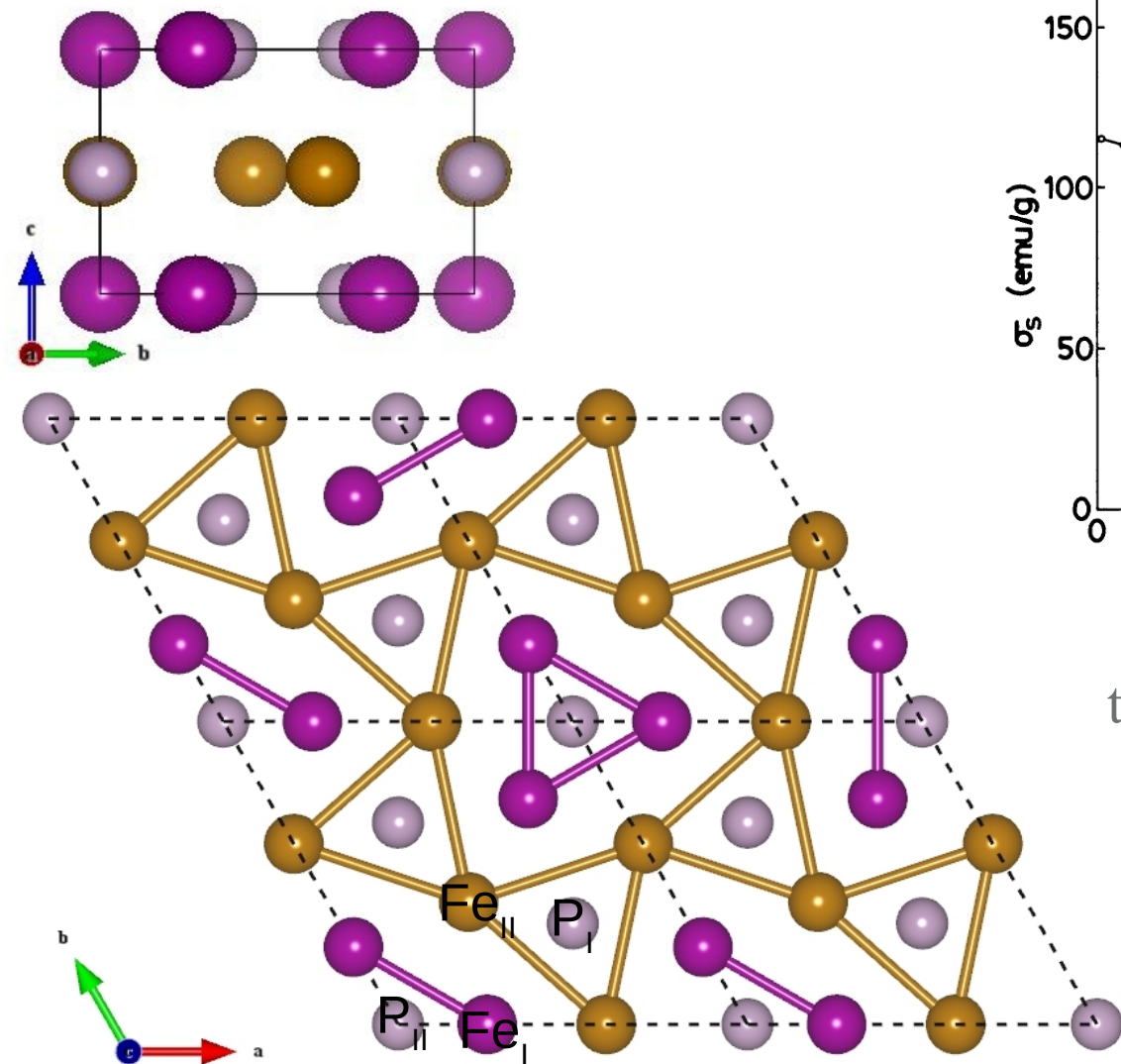


Parent \rightarrow Fe_2P

P-62m (189)



Fe_I	3f sites $\sim 0.8\mu_B$
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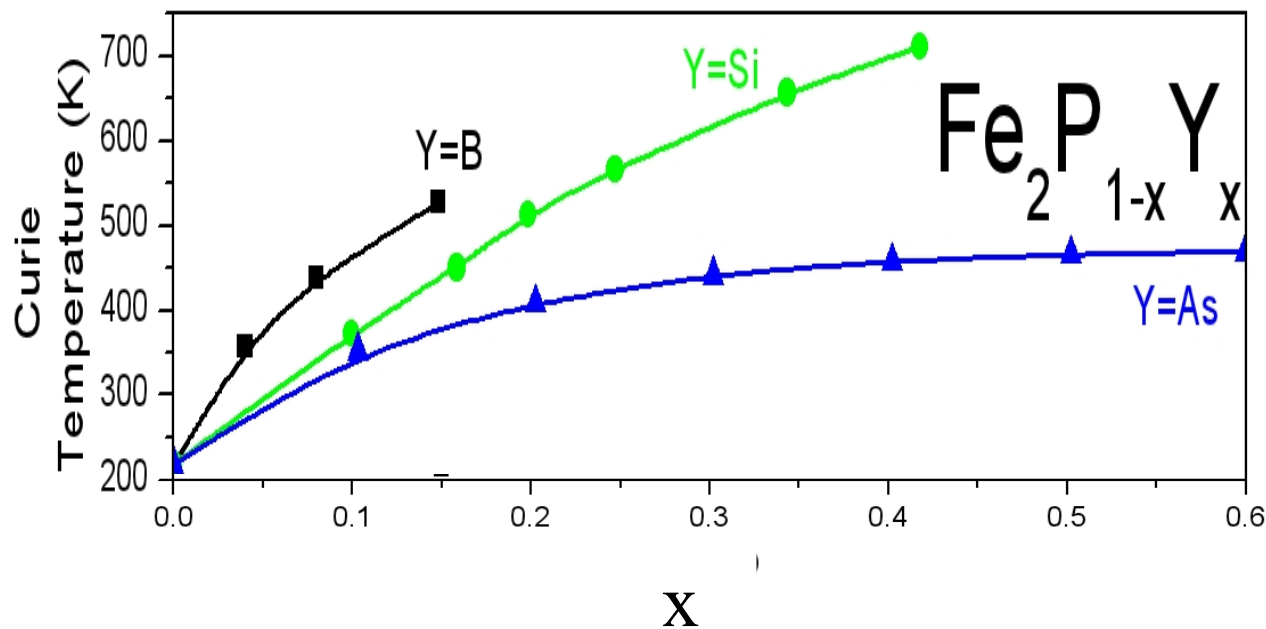
Only above ~ 700 K
the Curie-Weiss law is followed

local magnetic order above T_c !

H. Fujii et al. JPSJ 43, 1 (1977)

Magneto-elastic coupling in doped Fe_2P

Strong dependence of magnetic ordering temperature with doping



Data combined from:

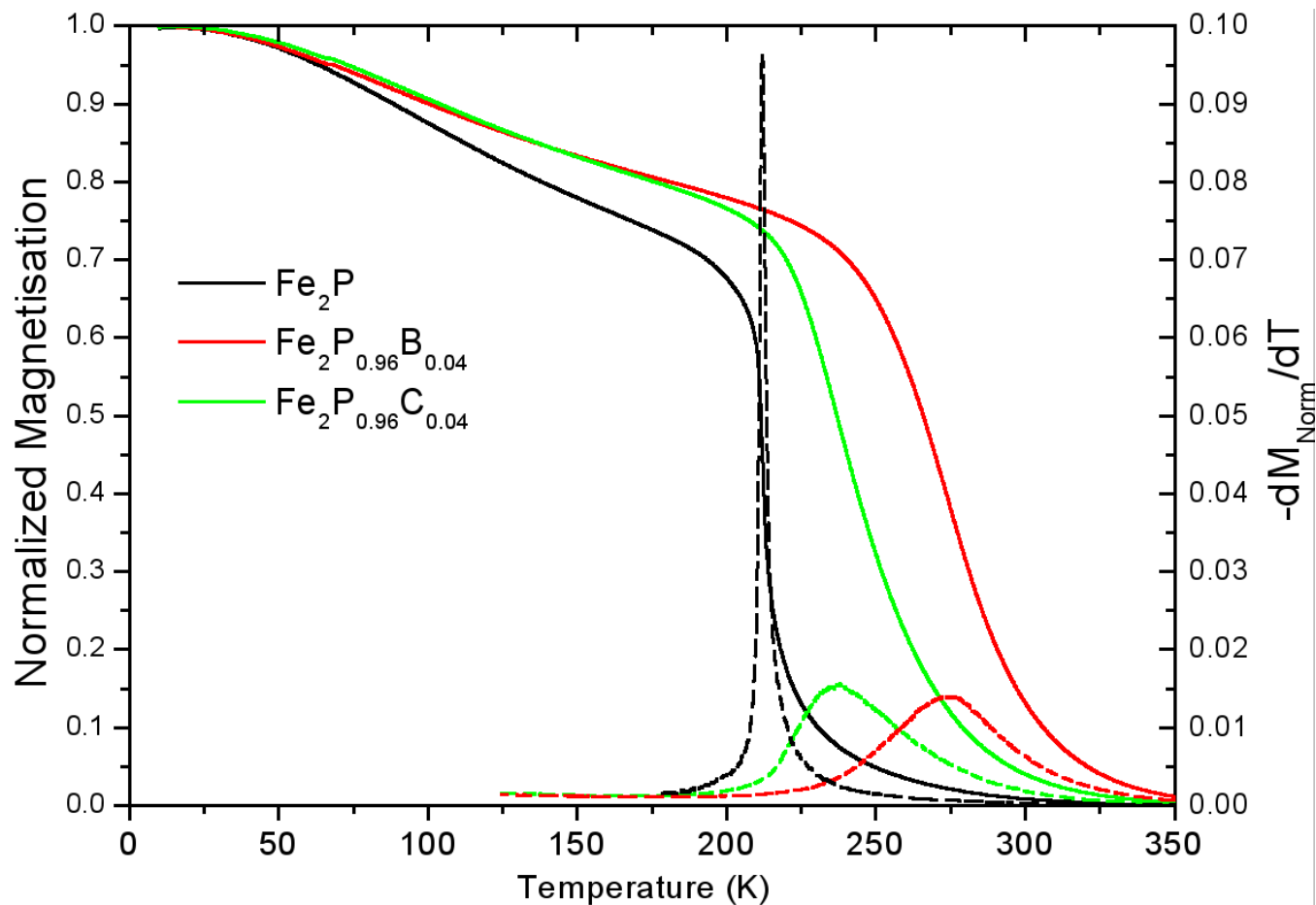
P. Jernberg et al. J. Sol. State Chem. 53 (1984)

R. Chandra et al. J. Sol. State Chem. 34 (1980)

A. Catalano et al. J. Sol. State Chem. 7 (1973)

Magnetic properties

Strong influence on doping with broadening transition



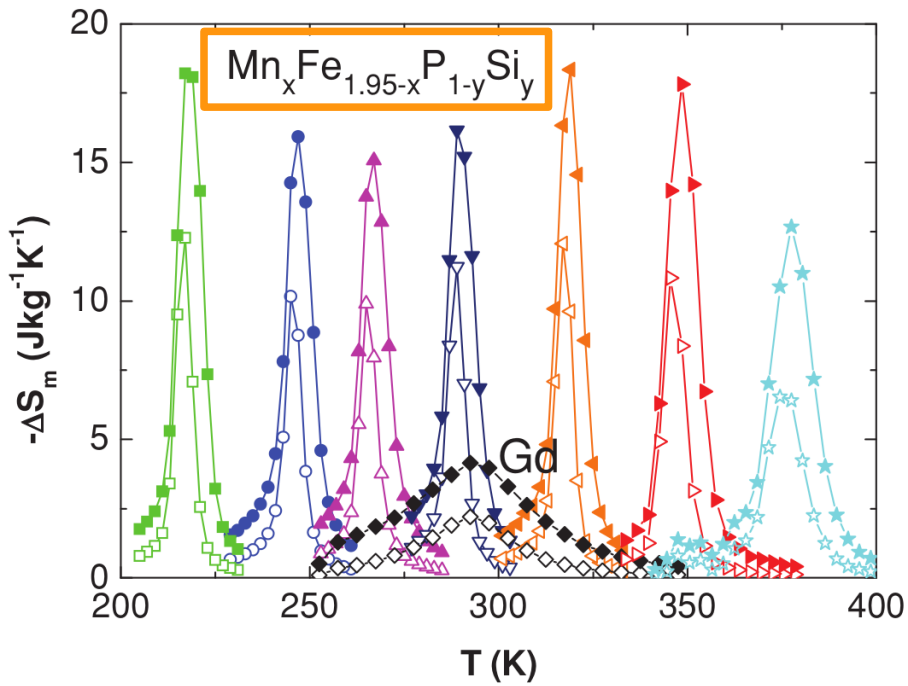
Magnetoelastic effects in doped Fe₂P
Z. Gercsi et al. Phys. Rev. B **88**, 024417 (2013)

MCE of Fe₂P single crystal
L. Caron et al.
PRB **88**, 094440 (2013)

Introduction

Metamagnetic transition in Fe₂P-based alloys

Tunable, field-dependent magnetic transition with large accompanying MCE



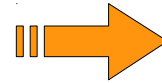
N. Dung et al.

Adv. Energy Mater. **1** 1215–1219 (2011)

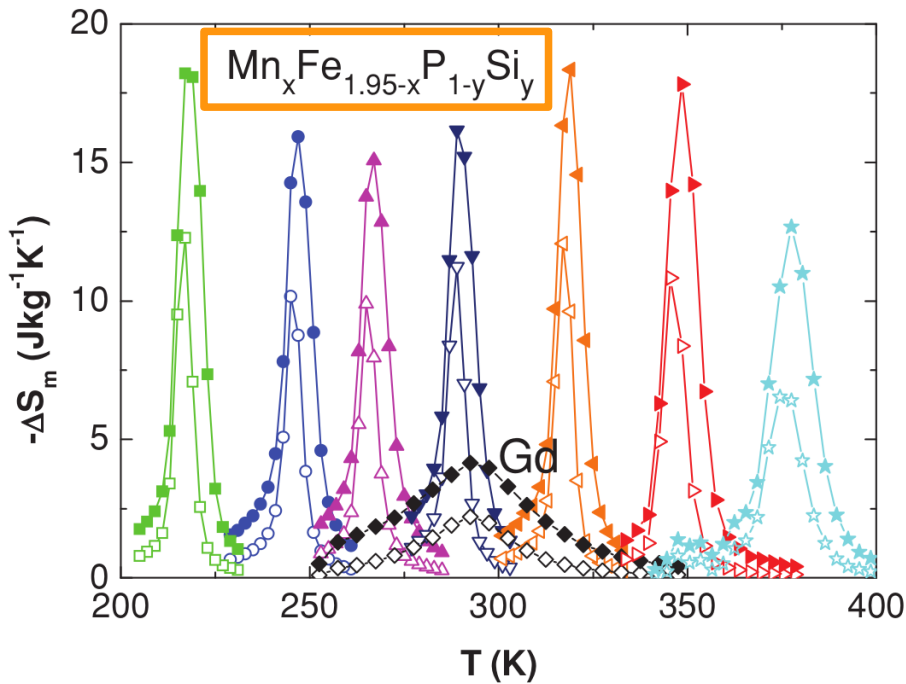
Introduction

Metamagnetic transition in Fe₂P-based alloys

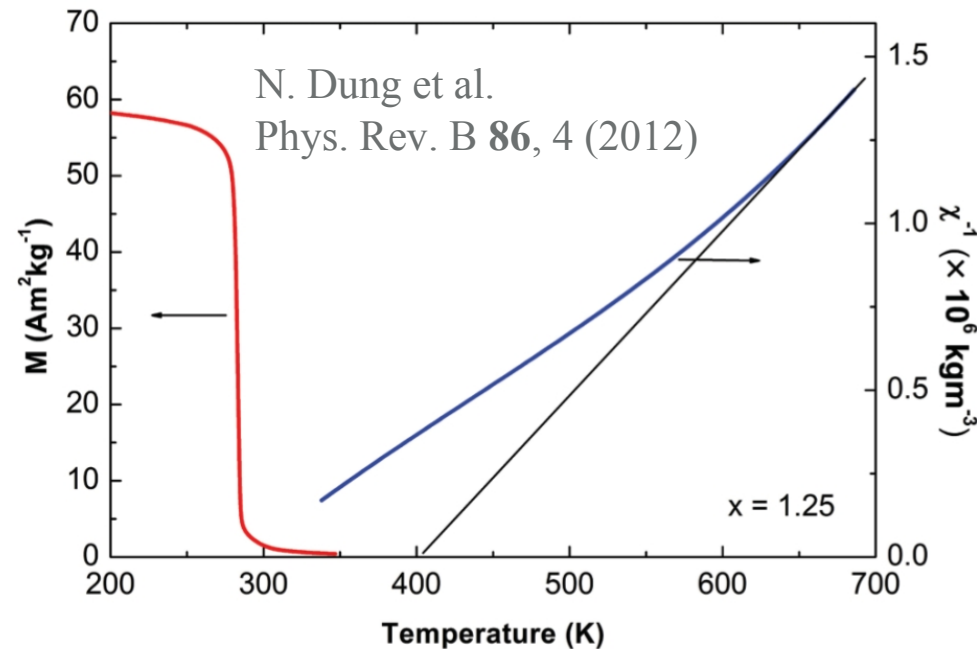
Tunable, field-dependent magnetic transition with large accompanying MCE



Similarly to parent alloy, short range magnetic order exist above T_c



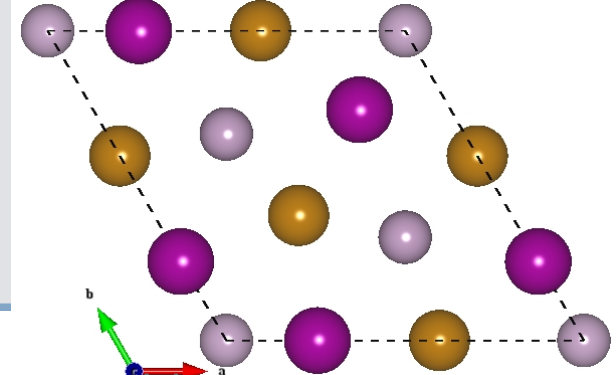
N. Dung et al.
Adv. Energy Mater. **1** 1215–1219 (2011)



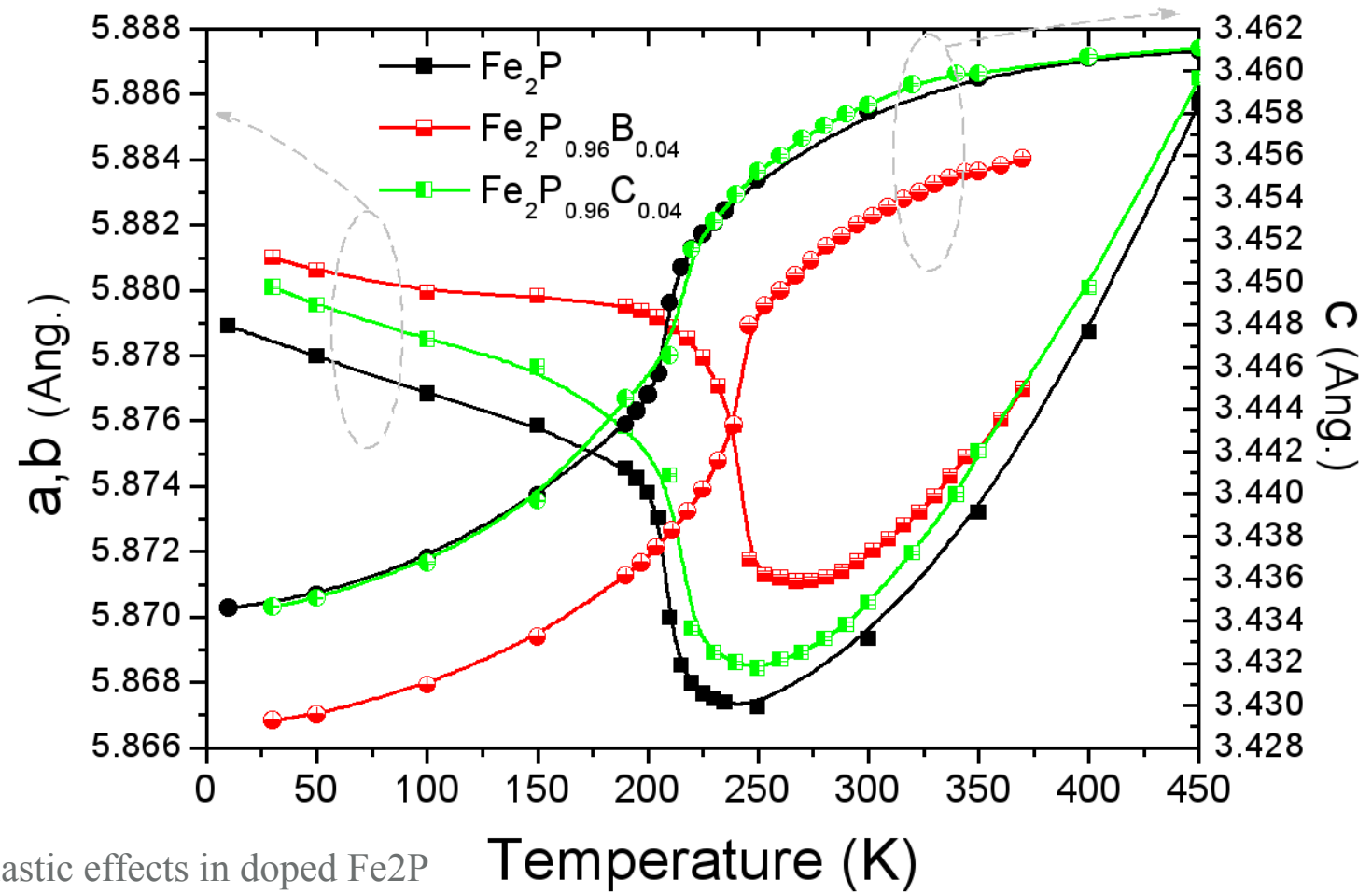
Also, Mn has a strong 3g-site preference as found experimentally.

Magneto-elastic response

Neutron diffraction (HRPD-ISIS)

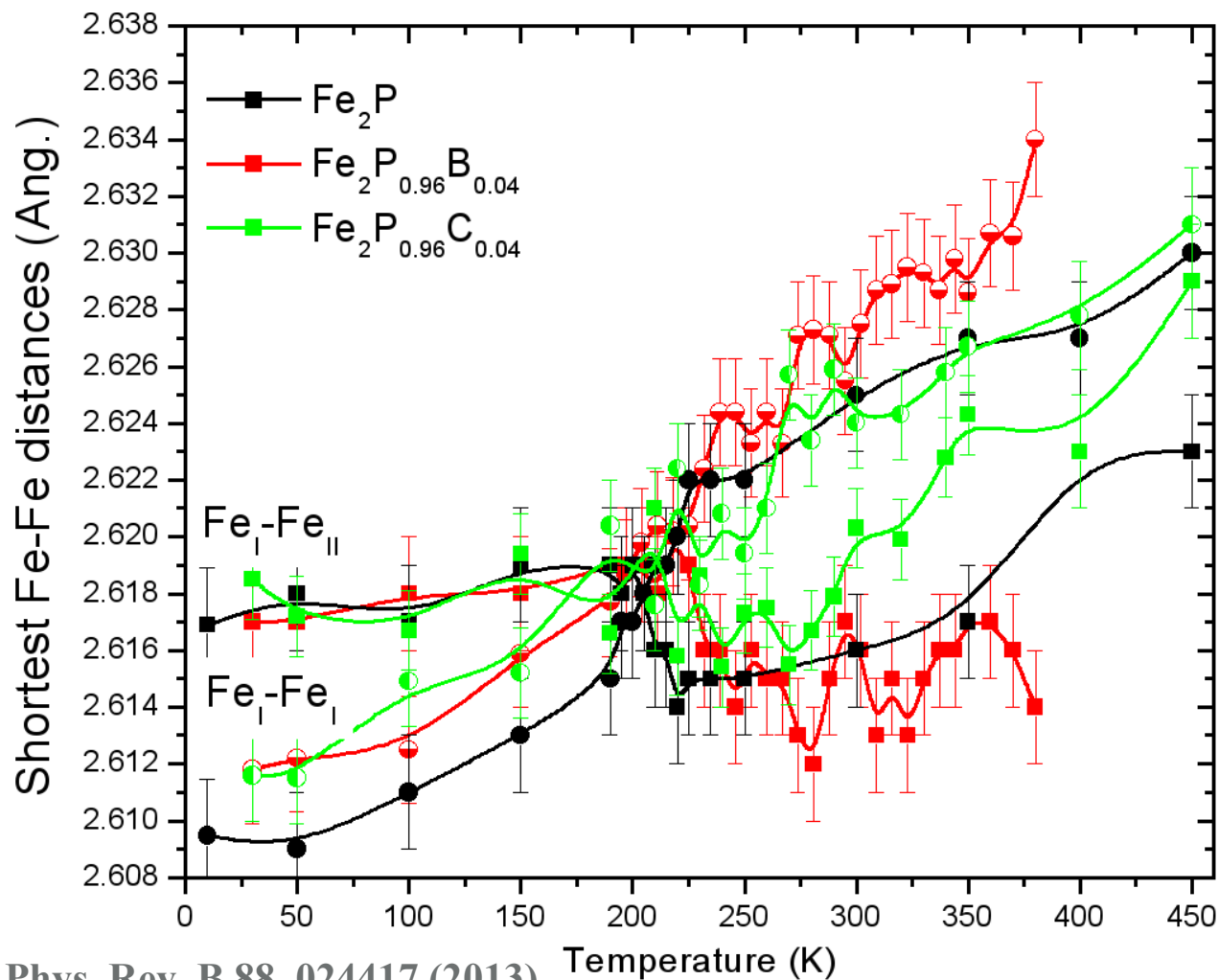
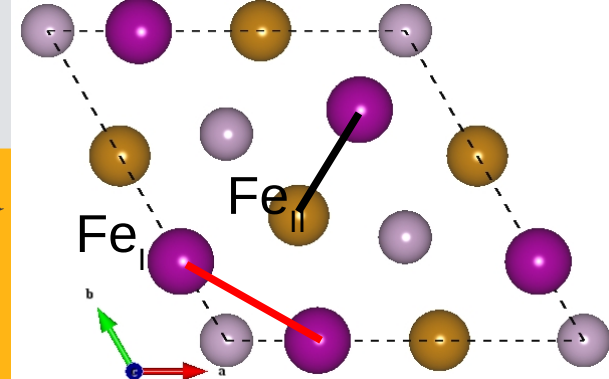


“Counteracting” change in lattice parameters over the transition.



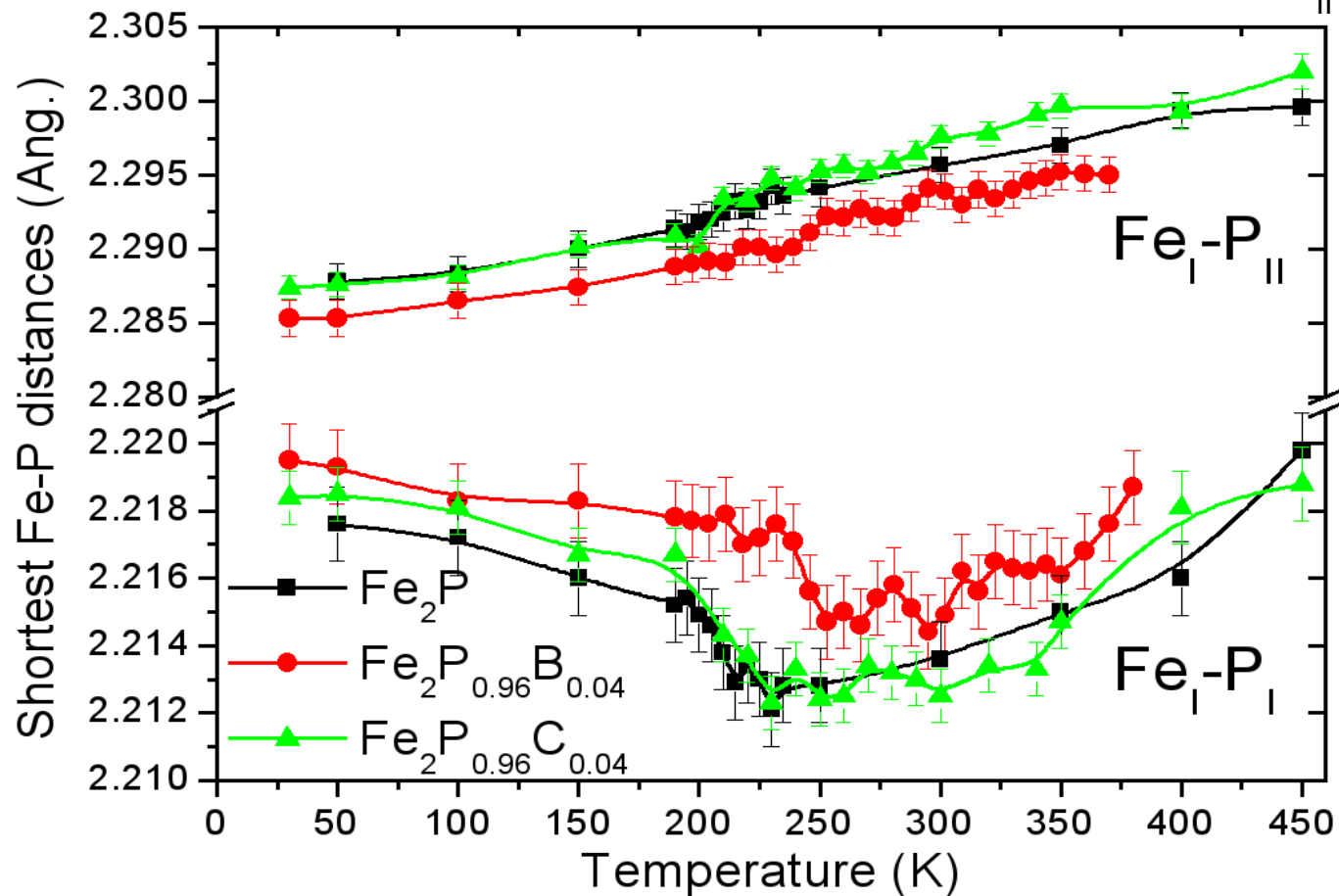
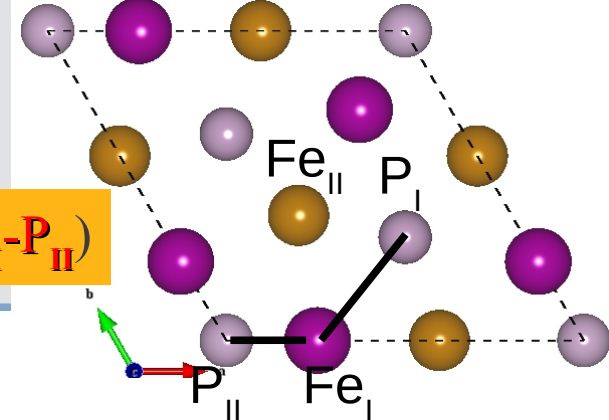
Magneto-elastic response

We observed an increase in the shortest $\text{Fe}_I\text{-Fe}_I$ distance and simultaneously a drop in the second shortest metal-metal distance ($\text{Fe}_I\text{-Fe}_{II}$) at the transition temperature.

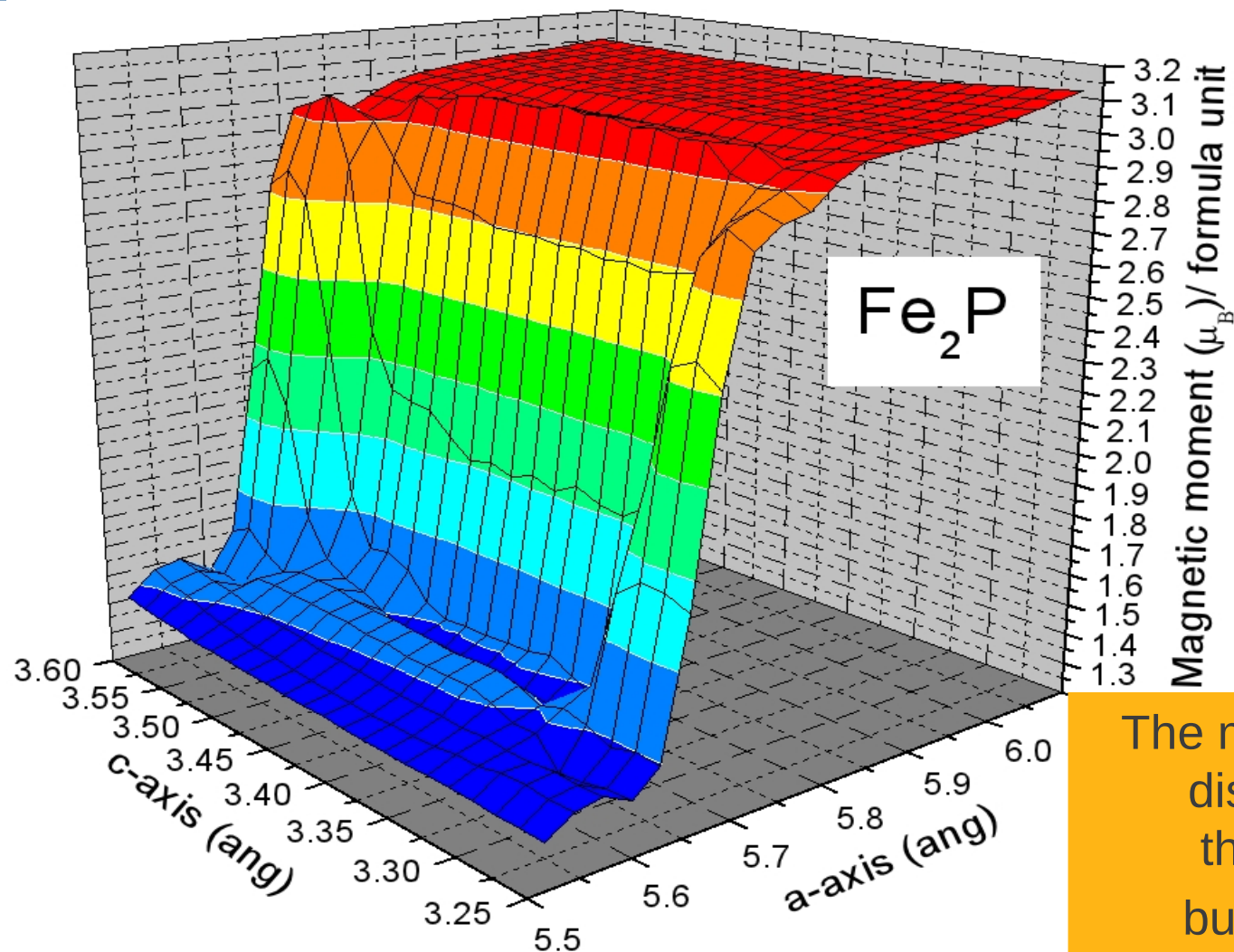


Magneto-elastic response

Change in the shortest metal-metalloid distances (Fe_I-P_I , Fe_I-P_{II})

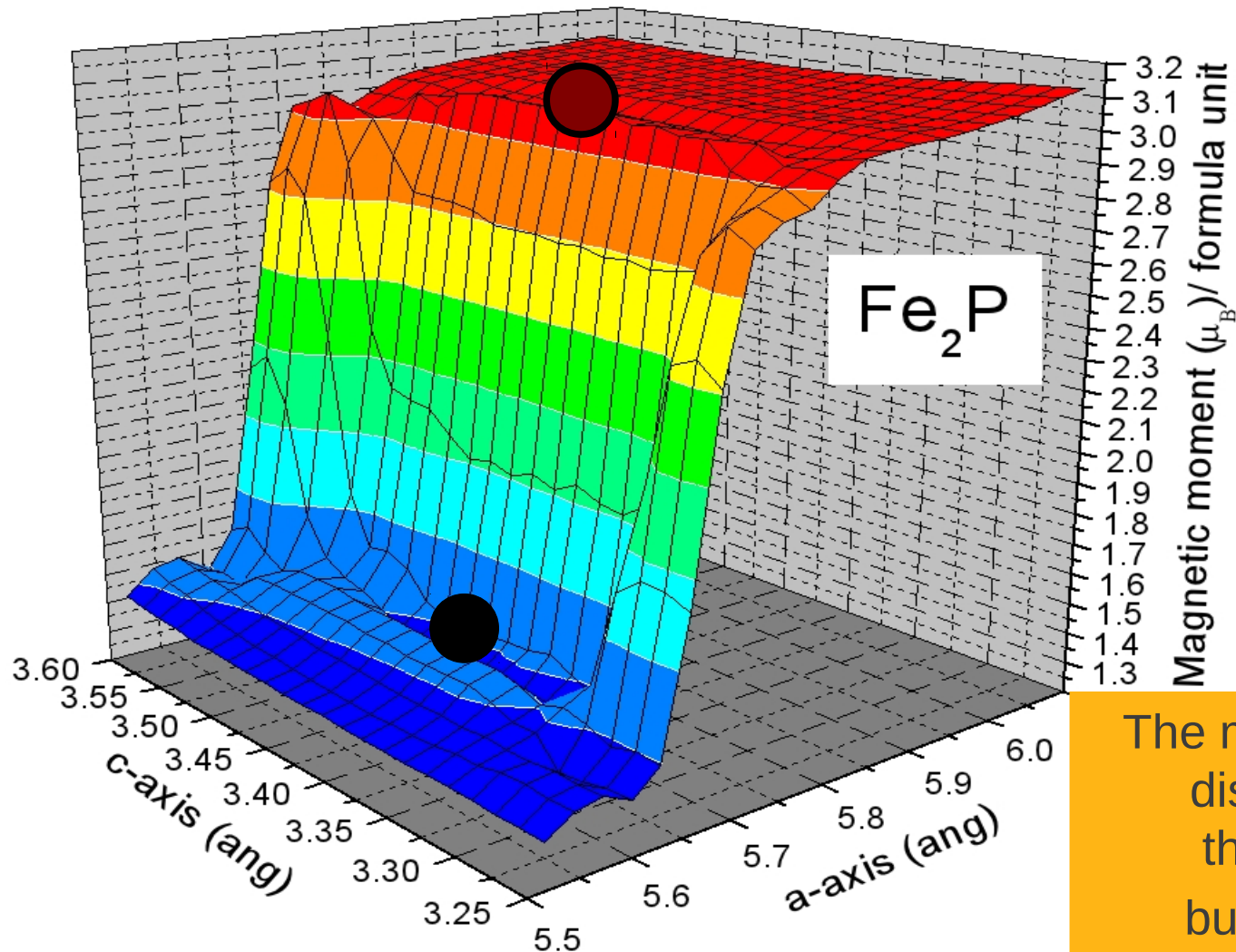


The calculations revealed → the basal plane is sensitive to the magnetic properties.



The magnetic moment disappears from the 3f-site (Fe_f) but only reduced on the 3g-site (Fe_{\parallel}) $\approx 1.4\mu_B$.

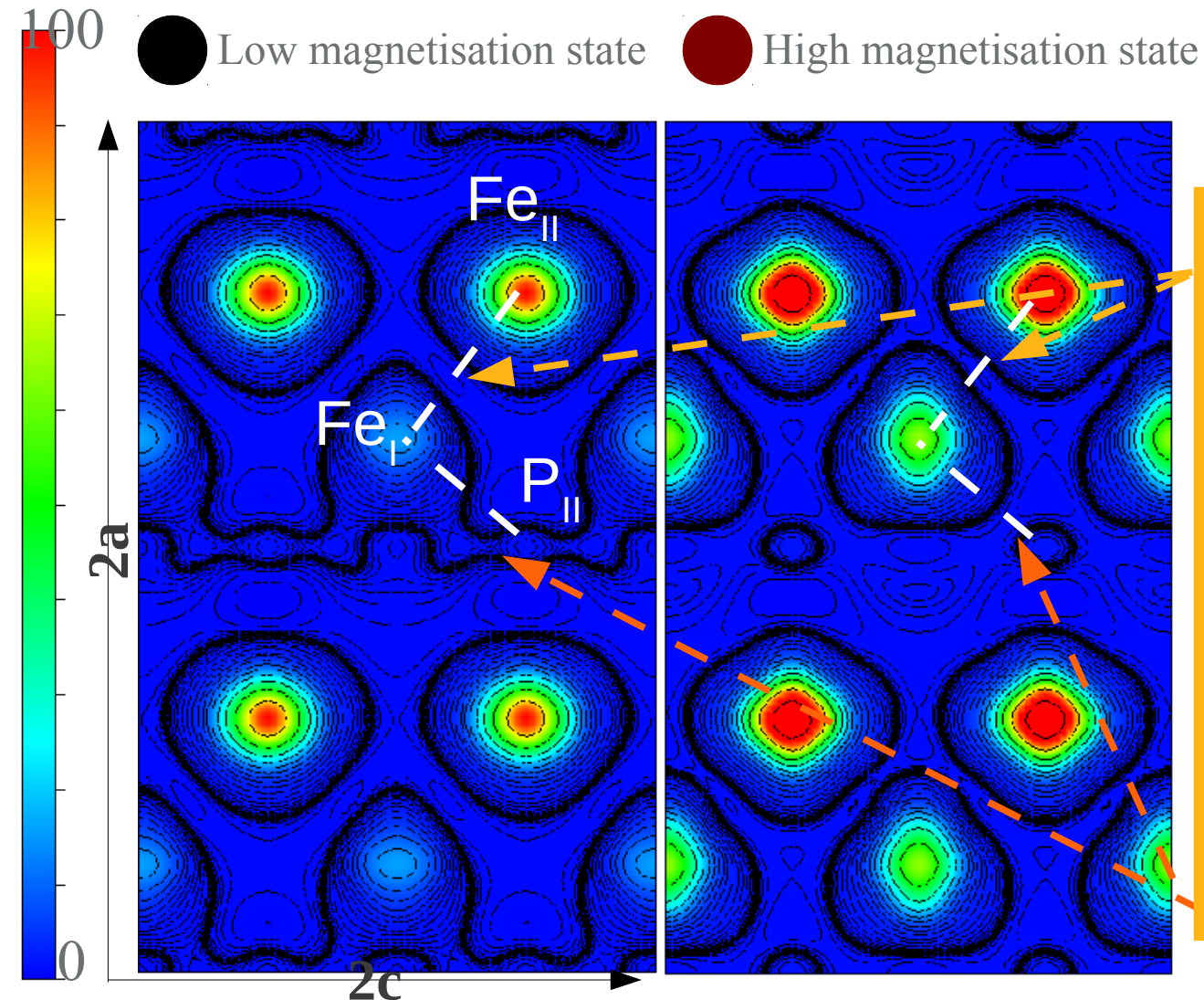
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Density Functional Theory

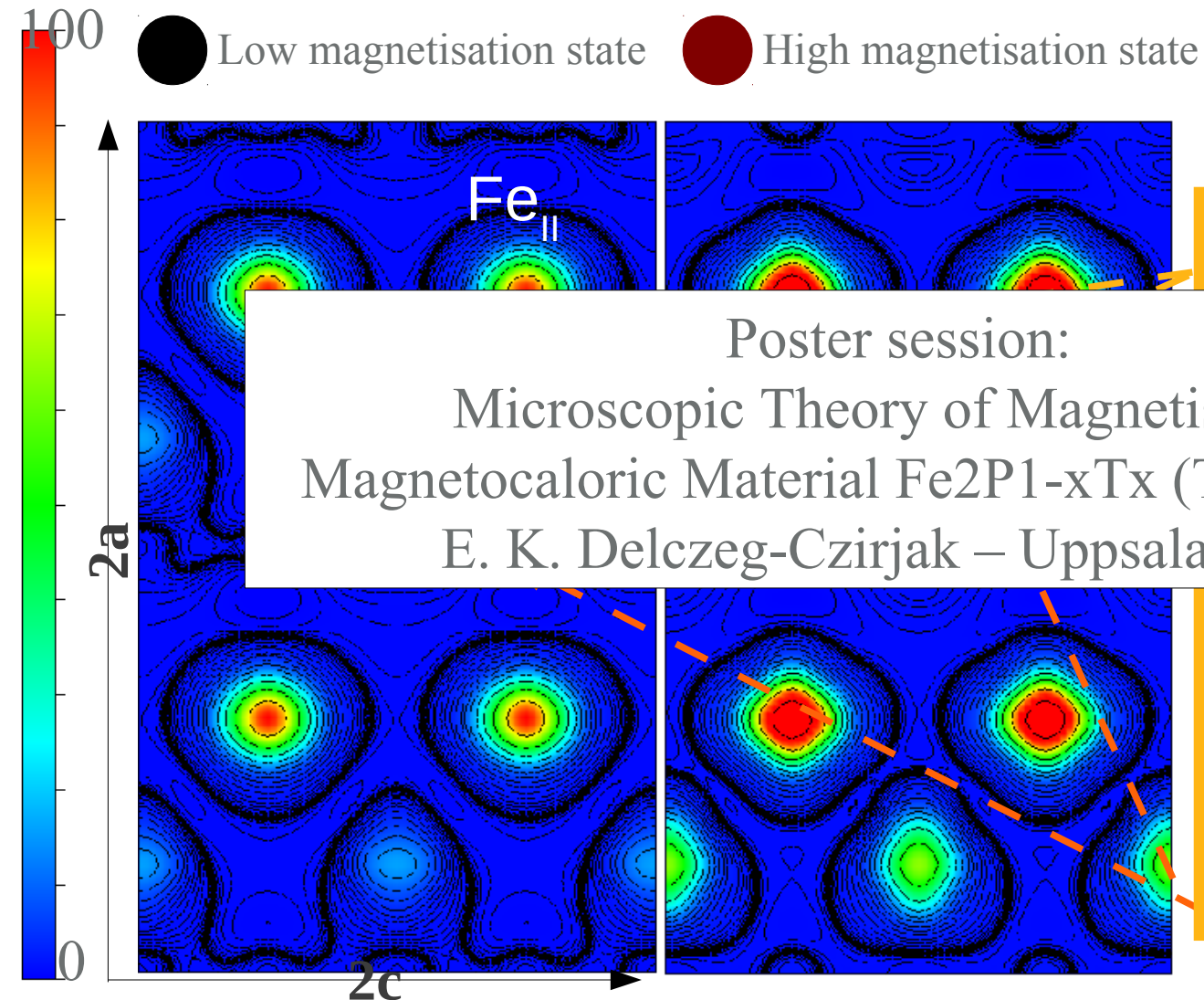
Magnetisation density plot in the a - c plane of Fe_2P



The closest metal-metal and metal-metalloid distances—the latter also linked to the metamagnetic 3f site—are strongly altered by both the d-d and p-d hybridization energies at the transition. As a result, the delocalization of the magnetization from the 3f site along the FeI - PII chains in the c -axis direction occurs, implying its strong influence on bonding.

Density Functional Theory

Magnetisation density plot in the a - c plane of Fe_2P



Poster session:
Microscopic Theory of Magnetism in
Magnetocaloric Material $\text{Fe}_2\text{P}_{1-x}\text{T}_x$ (T= B and Si)
E. K. Delczeg-Czirjak – Uppsala Univ.

The closest metal-metal
and metal-metalloid

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ite—are
both the
dization
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a result, the delocalization of
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chains in the c -axis direction
occurs, implying its strong
influence on bonding.

MMnX-based alloys

Hexagonal
(NiAs, Fe₂P, Ni₂In)

Orthorhombic
(Pnma)

Symmetry change
over magnetic
transition

(La,X)MnO₃
FM-AFM,CO,OO,SO

Ni₂MnGa
Symm change:
(Heusler to tetragonal)

Gd₅Ge₂Si₂
Symm change:
(Pnma to P1121/a)

La(Fe,Si)₁₃

Cubic Structure

(No symmetry change in transition)

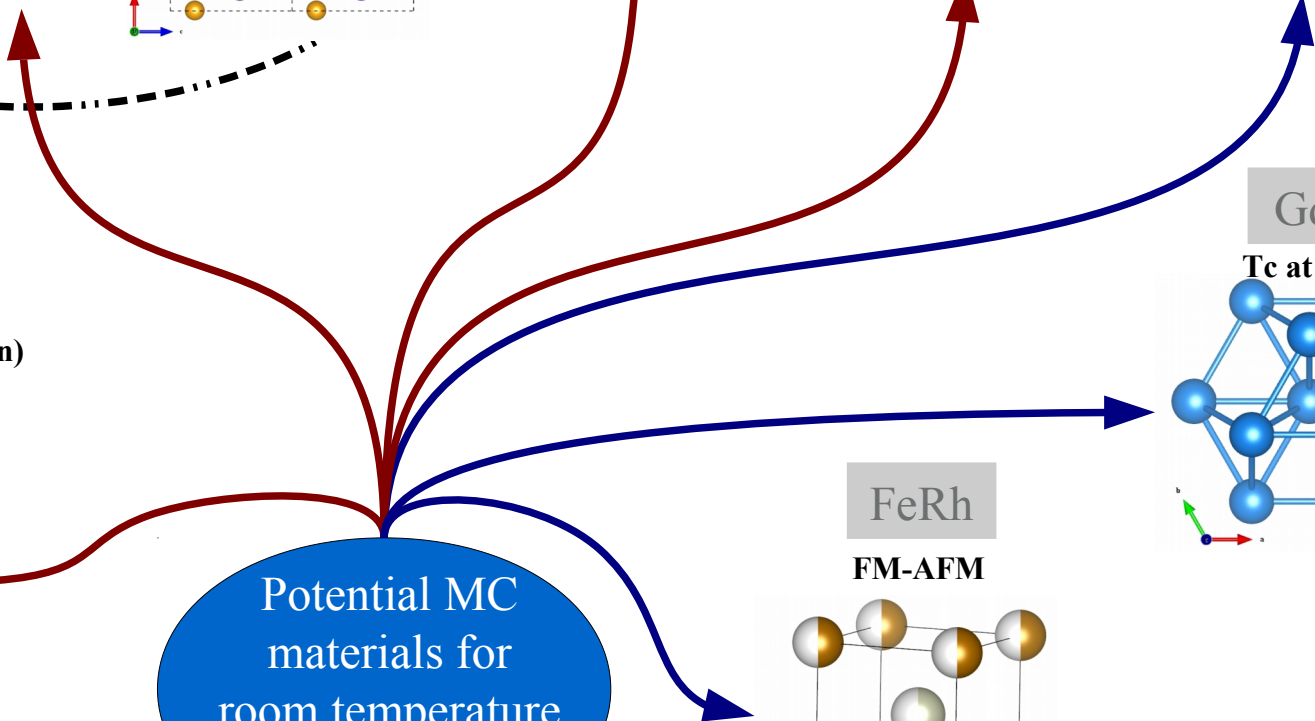
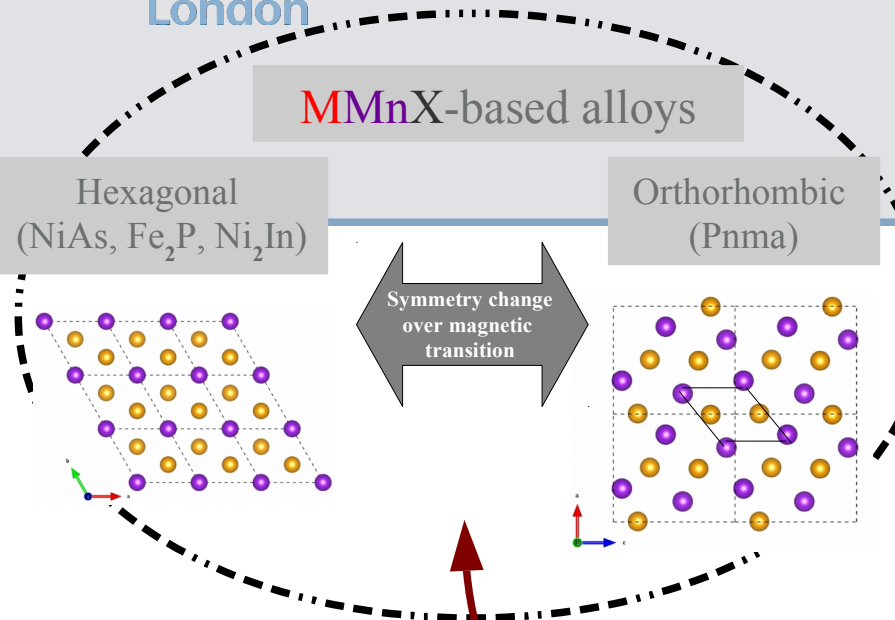
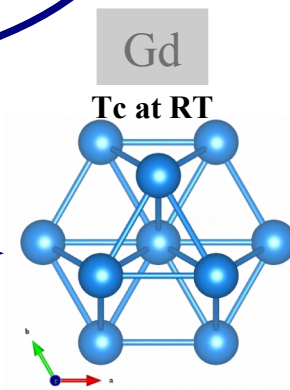
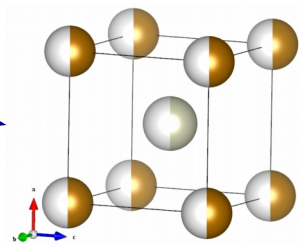
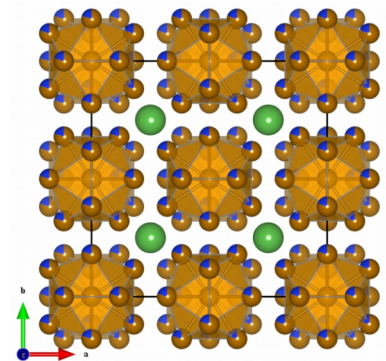
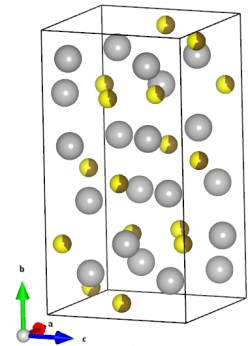
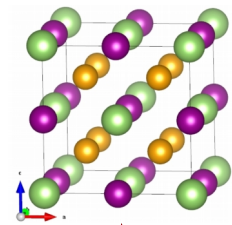
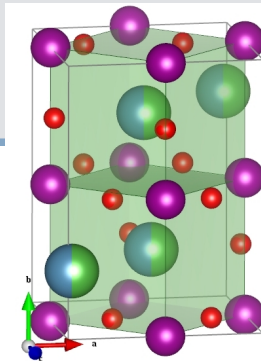
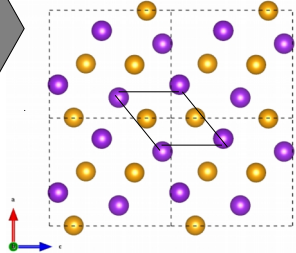
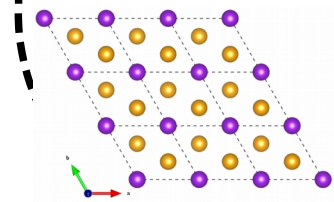
Potential MC
materials for
room temperature
refrigeration

FeRh

FM-AFM

Gd

Tc at RT



Thank you for your attention!

Acknowledgement

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- Kevin Knight, A. Daoud-Aladine, ISIS, UK
- K. Hono, NIMS, Japan

For further details, please see in

Z. Gercsi, E. K. Delczeg-Czirjak et al. Phys. Rev. B **88**, 024417 (2013)

E. K. Delczeg-Czirjak, Z. Gercsi et al. Phys. Rev. B **85**, 224435 (2012)

E. K. Delczeg-Czirjak, L. Bergqvist, O. Eriksson, Z. Gercsi et al. Phys. Rev. B **86**, 045126 (2012)

CoMnSi related works:

J. B. Staunton, M. dos Santos Dias, J. Peace, Z. Gercsi, and K. G. Sandeman Phys. Rev. B **87**, 060404 (2013)

A. Barcza, Z. Gercsi et al. Phys. Rev. B **87**, 064410 (2013)

Q. Recour, V. Ban, Z. Gercsi et al. Phys. Rev. B **88**, 054429 (2013)

Z. Gercsi, K. Hono and K.G. Sandeman Phys. Rev. B **83**, 174403 (2011)

Z. Gercsi and K.G. Sandeman Phys. Rev. B **81**, 224426 (2010)

A. Barcza, Z. Gercsi, K.S. Knight and K.G. Sandeman Phys. Rev. Lett. **104**, 247202 (2010)

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